

11 GRAPHICAL ABSTRACT



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13 ABSTRACT

A novel technique involving hydrothermal synthesis was employed to prepare nanocomposite precursors of lamellar double hydroxide based basic mixed oxides. This method extended ongoing research on iron-impregnated hydrotalcite-supported nanomaterials. The iron-based Fe/MgAl-h nanocomposite was prepared via dry impregnation, introducing a known % by weight of Fe (NO₃)₃.9H₂O metal salt into the hydrothermally synthesized MgAl-h structure. For comparative analysis, the photocatalytic performance of Fe/MgAl-h was compared with the coprecipitation-

synthesized Fe/MgAl-c after calcination at 400°C, using various characterization techniques. 20 Structural examination revealed a double-layered hydroxyl structure and the presence of iron oxide 21 phase Fe₂O₃ in Fe/MgAl-h400. Optical assessments indicated an indirect band gap energy of 1.89 22 eV for solid Fe/MgAl-h400, which is suitable for visible light absorption. Photocatalytic 23 experiments for methylene blue (MB) dye degradation in aqueous solution were performed under 24 artificial irradiation with a tungsten lamp as the visible light source, mirroring the conditions of 25 Fe/MgAl-c400. Using 50 mg/L MB concentration and 0.5 g/L catalyst quantity, increased MB 26 removal efficiency with time was observed. A comparative study of hydrothermal and 27 coprecipitation synthesis methods showed decreased photodegradation for the hydrothermally 28 prepared solid. Fe/MgAl-h400 and Fe/MgAl-c400 exhibited reaction efficiencies of approximately 29 46% and 67%, respectively, possibly attributed to their distinct indirect and direct band gap 30 characteristics. Moreover, a wider band gap in a semiconductor material offers several advantages 31 for the degradation of dyes. 32

Keywords: band gap, comparison, hydrothermal, co-precepitation, hydrotalcite, nanomaterials and
photocatalysis.

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36 Introduction

Layered double hydroxides, also known as anionic clays or hydrotalcite-like compounds, are indeed a class of solid basic catalysts that have offer a promising platform for the development of environmentally friendly and efficient catalytic processes in the production of fine chemicals and other chemical transformations. Researchers continue to explore and optimize their properties for various applications in green chemistry and sustainable chemical synthesis.

42 hydrotalcites represent a category of both man-made and naturally occurring inorganic solids. These
43 materials consist of two-dimensional layered structures composed of positively charged layers

resembling brucite (Mg(OH)2). Within these layers, you'll find bivalent and trivalent metal ions
arranged in an octahedral configuration (ALLMANN & HP, 1969; Taylor, 1973; Terzis *et al.*, 1987;
Pengfei *et al.*, 2020).

The hydrotalcite's general formula, denoted as $[M_{1-x}^{2+}M_x^{3+}(OH)_2]^{x+}[A_{x/n}^{n-} mH_2O]^x$, encompasses various metal cations, represented by "M," including both divalent (e.g., Mg²⁺, Zn²⁺, Ni²⁺, Co²⁺, Ca²⁺ etc.) and trivalent (e.g., Al³⁺, Fe³⁺, Cr³⁺, Mo³⁺, Co³⁺, Ga³⁺etc.) species. Additionally, "A" represents the intercalated anions. These hydrotalcite layers maintain electrical neutrality by balancing their positive charge with the presence of negatively charged interlayer anions (Mishra *et al.*, 2018b).

hydrotalcites, characterized by affordability, ready accessibility, and straightforward synthesis, have 53 emerged as highly promising options for a range of applications. This is due to their extensive 54 surface area, abundant surface-active sites, and impressive capacity for anion exchange, allowing 55 them to incorporate anions of various sizes and types within their layers. With the significant 56 attention hydrotalcites have garnered in recent years, a diverse array of hydrotalcite-based materials 57 has been developed through specialized techniques or the incorporation of various metal cations to 58 fulfill specific objectives (Fan et al., 2014; Peng et al., 2014; Patel et al., 2017; Arrabito et al., 59 2019; Yan et al., 2019; Ameena Shirin et al., 2021). 60

Over the past few decades, there has been a significant surge in interest surrounding the 61 development of nanoparticle-scale technology for a wide range of applications. As a result, various 62 synthesis techniques have emerged, encompassing gas phase, liquid phase, two-phase, sol-gel 63 methods, co-precipitation, and hydrothermal methods (Miyata & Kumura, 1973; Reichle, 1985; 64 Cavani et al., 1991; Sels et al., 2001). In this study, we aim to compare two prominent methods, 65 namely the co-precipitation method, which involves maintaining both constant and variable pH 66 levels (Crepaldi et al., 2000), and the hydrothermal method (Reichle, 1986). It's worth noting that 67 hydrothermal synthesis can be combined with co-precipitation to enhance the crystallization of 68

69 hydrotalcite, a process with distinct advantages and limitations. Additionally, it's important to 70 recognize that the synthesis method employed significantly influences the structure of LDH, with 71 co-precipitation being a widely utilized approach in the production of hydrotalcite materials.

The precipitation at constant pH is generally applied to synthesize the hydrotalcites structure. The 72 synthesis is carried out by the addition of the solution of two different metal salts dissolved in the 73 distilled water. Which is added drop wise over an aqueous solution containing the mixture of anion 74 to be intercalated and NaOH to accelerate coprecipitation reaction in the pH range. The reaction is 75 carried out at room temperature (RT). The precipitate is then filtered and washed several times with 76 distilled water. The drying temperature is not allowed to exceed 120 °C (Terzis et al., 1987). The 77 78 use of this method leads to obtaining of pure and homogenous hydrotalcite with a high crystallinity. The advantage obtained by the co-precipitation at constant pH lies on the textural properties of the 79 hydrotalcites materials. The hydrotalcites structure prepared by this method bears a small particle 80 81 size, a high specific surface area and a high average pore diameter than the materials prepared at variable pH (Faour et al., 2010; Qin et al., 2020). On the other hand, the co-precipitation method 82 allows to finely tuning the structure of the synthesized materials by controlling the MII/MIII molar 83 ratios, type of interlayer anion, the synthesis time, temperature, and the pH (Thevenot et al., 1989; 84 Zhao et al., 2003; Kloprogge et al., 2004; Klemkaite et al., 2011). 85

The hydrothermal synthesis is the combination with co-precipitation method to improve 86 hydrotalcite crystallization, were consists of the treatment of fresh precipitate of co-precipitation 87 synthesis resulting of mixing the solution containing metal salts and alkaline solution. The mixtures 88 are usually put inside a Teflon-lined autoclave in the temperatures range from 80 to 350°C and the 89 reaction times varied from hours to days (Miyata, 1980). The advantages of this method include its 90 simplicity and transformation of small crystallites of hydrotalcite into larger ones (Cavalcanti et al., 91 1987). The Mg-Al prepared by this method had hexagonally shaped crystallites with a lateral 92 dimension of 1-3 µm (Xu & Lu, 2005). This technique is recommended to prepare mixed metal 93

94 oxides from hydrotalcite precursor (Xu & Lu, 2005; Jing *et al.*, 2010), ternary hydrotalcites systems
95 and inorganic/organic hybrid hydrotalcite (Carja *et al.*, 2008; Mishra *et al.*, 2018a).

Recently, research on hydrotalcite has been directed towards the development of new hybrid or 96 composite systems with improved physicochemical properties. The lamellar structures, in spite of 97 their remarkable basic properties (Fan et al., 2014; Patel et al., 2017; Mishra et al., 2018b), are less 98 used in catalytic reactions than the mixed oxides, which are easier to process and exhibit higher 99 specific surface areas. Therefore, it will be very useful to adapt the basic properties of mixed oxides 100 obtained from hydrotalcite precursors which will be synthesized by important methods like 101 coprecipitation and hydrothermal method in order to carry out reactions requiring different basic 102 103 resistances.

Numerous hydrotalcite supported nanoparticles, two-dimensional graphene oxide sheets or metal
complexes have already proven to be very effective towards catalysis, photocatlysis, adsorption,
sensing and biological applications (Choudary *et al.*, 2002; Garcia-Gallastegui *et al.*, 2012; Islam *et al.*, 2015, 2019; Allou *et al.*, 2017; Asif *et al.*, 2017). Also, it is noteworthy that iron Oxide
Nanoparticles supported on clays exhibit many interesting properties that can beused in a variety of
applications (Oliveira *et al.*, 2003; Nishio *et al.*, 2007; da Silva *et al.*, 2010; Chen *et al.*, 2017;
Kerchich *et al.*, 2021).

In this article, Kerchich et al. (2021) (Kerchich *et al.*, 2021), conducted a study aiming to provide a systematic overview and a concise comparison between hydrothermal synthesis and co-precipitation methods. The research also includes a structural analysis and explores the potential application of Fe/MgAl in photocatalysis for the purpose of removing methylene blue dye.

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116 Experimental section

117 Synthetic methods of semiconductor nanoparticles

118 Preparation of MgAl-h support by hydrothermal method

The MgAl support was prepared using two distinct methods. In the first method, known as the 119 hydrothermal method, the reaction involved the addition of appropriate quantities of Mg 120 (NO₃)_{2.6H₂O (0.2 M, Aldrich, 99%), Al (NO₃)_{3.6H₂O (0.1 M, Aldrich, 99.1%), and NaOH (2 M,}} 121 Aldrich, 99%) to achieve a constant pH of approximately 12. A laboratory pH meter (Hanna HI 122 2211) was employed to control the addition of the alkaline solution. The resulting mixture was 123 placed inside a Teflon-lined autoclave and maintained at a temperature of 80°C for 17 hours 124 (overnight). Subsequently, the formed gel underwent a washing process using distilled water and 125 was finally dried overnight (Chebout et al., 2010; Kerchiche, Chebout, & Bachari, 2017; Riman, 126 Suchanek, & Lencka, 2002). 127

On the other hand, in the coprecipitation method used to synthesize MgAl-c material with the same 128 Mg/Al, the initial step paralleled the hydrothermal method, involving the addition of the specified 129 amounts of Mg (NO₃)₂.6H₂O (0.2 M, Aldrich, 99%), Al (NO₃)₃.6H₂O (0.1 M, Aldrich, 99.1%), and 130 NaOH (2 M, Aldrich, 99%) to maintain a constant pH of around 12. The subsequent step diverged 131 as the mixture was transferred to a balloon heater under nitrogen reflux, with the temperature 132 maintained at 70°C for 18 hours to achieve a pure, well-crystallized compound. The final product 133 was obtained after washing/centrifugation with bidistilled water and subsequent drying in an oven 134 at 80°C for 12 hours (Riman et al., 2002; Chebout et al., 2010; Kerchiche et al., 2017). Finaly, both 135 136 based HDL materials underwent a crushing and sieving process to achieve a fine particle size before before appropriate use. The host material resulting from the hydrothermal method was denoted as 137 MgAl-LDHh, while the material synthesized through coprecipitation was designated as MgAl-138 LDHc. 139

140 Preparation of nanocomposite Fe₂O₃/MgAl-h

The protocol for preparing the Fe/MgAl solid by impregnation consists in bringing the MgAl support into contact with an iron content of a known % by weight of the metal precursor, the metal salt Fe(NO₃)₃.9H₂O (Aldrich, 97%). For this work, the choice is focused on the dry impregnation 144 method to prepare the Fe/ MgAl materials. The volume of the metal salt solution which must 145 correspond to the volume of the pores of the support. This method is limited by the solubility of the 146 precursor metal and is used to prepare catalysts with weak interactions between the metal and the 147 support (Desportes *et al.*, 2005). The mixture is then stirred for 2 hours by magnetic stirring and the 148 excess solution is removed by evaporation then by drying in an oven overnight at 80 °C and 149 calcined at 400 °C for 4 hours. The materials were denoted Fe/MgAl-h and Fe/MgAlc for 150 hydrothermal and coprecipitation method respectly.

151 Characterization techniques

Diffuse reflectance infrared Fourier-transform (DRIFT) spectra of the samples were obtained using 152 a Perkin-Elmer FTIR 1000 spectrometer, covering the wavelength range from 400 to 4000 cm⁻¹. 153 The chemical composition of the materials was determined using a Rigaku "ZSX Primus II" X-Ray 154 Fluorescence Spectrometer. Thermogravimetric (TGA) and differential thermal analysis (DTA) 155 were conducted using a Thermal Analyzer (Model SDT Q 600-TA) instrument. The sample 156 underwent heating from ambient temperature to 900 °C under N2 flow, with a heating rate of 10 °C 157 min⁻¹. X-ray powder diffraction (XRD) patterns were recorded utilizing a Bruker D8 Advance X-158 ray diffractometer, employing Cu Ka1 radiation ($\lambda \alpha = 1.54184$ Å, 40 kV, and 50 mA). The data 159 were collected within the 7° to 80° 2 θ range. The structural characteristics of the materials were 160 observed through scanning electron microscopy (SEM) using a FEI Quanta200 microscope 161 operating at 20 kV. UV-vis diffuse reflectance (DRS) spectra were recorded using a Shimadzu 162 model UV-2100 spectrophotometer equipped with an integrating sphere accessory, with BaSO4 163 serving as the reference material. The obtained reflectance (R%) from the DRS spectrum was 164 utilized to determine the Eg value. The resulting solution was subject to analysis via atomic 165 absorption spectroscopy (240-FS Fast Sequential, AA range from Agilent). 166

167 **Photocatalytic reactivity study**

168 To investigate the photocatalytic degradation of the dye, we followed a specific experimental

169 procedure. This involved preparing a solution with a fixed concentration of the methylene blue (MB) and a known quantity of synthetic material. We conducted these experiments in a 500 mL 170 Pyrex photochemical reactor with a double-wall design to maintain a constant temperature of 25°C, 171 achieved by employing a temperature-controlled bath. The light source used for irradiation was a 172 200 W tungsten lamp emitting light in the range of 400–800 nm, with an intensity of 2.10×10^{19} 173 photons per second. After the photocatalytic reaction, the suspension was subjected to 174 centrifugation at 3000 rpm for 10 minutes. Subsequently, we filtered the suspension to eliminate 175 any catalyst particles before performing absorbance measurements. To quantitatively assess the 176 results, we measured the absorbance at the wavelength $\lambda max = 665$ nm using a Varian UV-visible 177 Spectrophotometer (Cary 50 UV-vis Spectrophotometer). 178

179 Results and discussion

180 Structure and composition

All the synthesized hydrotalcites exhibited two-step decomposition upon heating under air (Fig. 1). 181 Initially, the first step was observed at temperatures ranging from 220 to 250 °C, resulting in a 15% 182 reduction in weight. This initial weight loss corresponded to the removal of interlamellar water 183 molecules (Riman et al., 2002; Desportes et al., 2005; Kerchiche et al., 2017; Ramos-Ramírez et 184 al., 2020). Subsequently, in the temperature range between 250 and 600 °C, a weight loss of 185 approximately 25% and 35% was observed for MgAl-h and Fe₂O₃/MgAl-h, respectively. These 186 weight reductions were attributed to two processes: the decomposition of interlamellar carbonate 187 and the dehydroxylation of the hydrotalcite sheets (Ramos-Ramírez et al., 2019). In this 188 189 temperature range, the lamellar structure underwent distortion due to the formation of mixed oxides (Rey et al., 1992). This distortion led to the emergence of exothermic peaks at approximately 350 190 and 550 °C, corresponding to the dehydroxylation of the brucite-like layers and the decomposition 191 of incorporated anions, as explained by (Wang et al., 2007). 192

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Fig.1. ATG/DTG of non-calcined MgAl-h and Fe/MgAl-h

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The infrared spectra presented in Figure 2 for our samples exhibit typical characteristics associated 205 with lamellar double hydroxides, consistent with findings from previous studies on hydrotalcites 206 (Fernández et al., 1998; Das et al., 2002b; Ferreira et al., 2004). To know that the spectral 207 characteristics of Fe/MgAl closely match those documented in prior literature, as evidenced by the 208 findings of (Riman et al., 2002; Chebout et al., 2010; Mamat et al., 2014; Kerchiche et al., 209 2017). These spectra reveal similar vibration bands. Specifically, a broad band observed between 210 211 3450 - 3500 cm-1 is attributed to the presence of hydroxide groups (OH) within the layers and to intercalated and/or adsorbed water molecules. Another vibration band in the range of 1630-1640 212 cm-1 can be ascribed to the deformation of intercalated water molecules HOH (δHOH) (Leofanti et 213 al., 1998). 214

The main bands associated with intercalated and adsorbed anions are observed within the 2000 to 215 1000 cm⁻¹ range (Cavani et al., 1991). Furthermore, the remaining bands below 1000 cm⁻¹ 216 correspond to vibrations of M-O-M, which are characteristic of the double lamellar structure (Das 217 et al., 2002b). Notably, a persistent vibration at 1450 cm⁻¹, observed in the material Fe/MgA1-h400, 218

is indicative of the presence of maghemite iron oxide (γ -Fe₂O₃). In Figure 2, it is evident that the peak intensities related to water dehydration and dehydroxylation decrease, a phenomenon attributed to the calcination of the materials at 400 °C(Barakat, 2012; Kerchich *et al.*, 2021).

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the elemental analyses of the samples are reported in Table.1, shown that the experimental molar ratios $M^{2+/}M^{3+}$ are in good agreement with the theoretical values ($M^{2+/}M^{3+} = 2$). On the other hand, $M^{3+/}(M^{2++}M^{3+})$ values are between 0.31 and 0.32, these values correspond to the optimal ratios necessary for achieving a well-formed crystalline LDH structure (Miyata, 1983; Fernández *et al.*, 1998; Leofanti *et al.*, 1998; Das *et al.*, 2002a; Barakat, 2012; Mamat *et al.*, 2014).



Samples	T ℃	Formulas			x =	%	Eg
			${ m M}^{2+}/{ m M}^{3+})_{theo}{ m M}^{2+}/{ m M}^{3+})_{ex}$		M ³⁺ /(M ²⁺⁺	Fe	(eV)
				р	M ³⁺)		
MgAl-h	400	Mg _{0.68} Al _{0.32} -h	2	2.1	0.32	-	-
Fe ₂ O ₃ /MgAl-h	400	Fe/Mg _{0.69} Al _{0.31} -h	2	2.22	0.31	7.73	1.89
Fe ₂ O ₃ /MgAl-c	400	Fe/Mg _{0.68} Al _{0.32} -c	2	2.06	0.32	9.73	2.02

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The X-ray diffraction (XRD) patterns presented in Figure 3 offer valuable insights into the samples 238 we prepared. These patterns exhibit distinctive characteristics for various samples, such as non-239 calcined samples Mg-Al-h and Mg-Al-c, as well as calcined samples such as Mg-Al-h400, Mg-Al-240 c400, Fe/Mg-Al-h400, and Fe/Mg-Al-c400. Notably, these patterns reveal prominent diffraction 241 peaks at specific 20 values (11.87, 23.75, 34.76, 39.1, 46.49, 61.12, 62.26, 66.32, 72.48, and 75.08°) 242 corresponding to the (003), (006), (009), (012), (018), (110), and (113) planes. These peaks 243 unequivocally confirm the presence of the MgAl structure, consistent with data from PDF-#38-244 0487, which describes a rhombohedral structure (Chen & Qu, 2003; Lakraimi et al., 2006; Hong et 245 al., 2014). However, it's important to note that the absence of peaks corresponding to (003) and 246 (006) after heat treatment indicates the disintegration of the hydrotalcites lamellar structure, 247 aligning with previous research findings (Chebout et al., 2010; Kerchiche et al., 2017; Ramos-248 Ramírez et al., 2020). This transformation suggests a shift from the hydrotalcite structure to oxides. 249 Remarkably, new peaks emerge at 20 values of 31.74, 35.35, 42.91, 62.40, 74.89, and 78.60°, 250 indicating the presence of MgO (JCPDS No. 87-0653). In contrast, for Fe/MgAl, diffraction peaks 251 appear at 20 values of 22.9, 29.1, 31.6, 39.0, 48.3, 55.6, and 56.6°, clearly indicating the presence of 252 iron oxide in the γ -phase (γ -Fe2O3, maghemite). The corresponding interlayer spacing (d003), 253 approximately 0.8 nm for all three samples, is consistent with a Mg/Al molar ratio close to 2 in the 254 brucite-like layers and the presence of carbonate as charge compensating anions, as reported in 255

previous studies (Wang *et al.*, 2007; Chebout *et al.*, 2010; Kerchiche *et al.*, 2017).







Fe/MgAl-c400

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The "a" parameter, representing the average metal-to-metal distance within the layers and determined from the (110) diffraction line, was found to be 0.3 nm for the MgAl sample and 0.2 nm for the Fe/MgAl sample. This observation substantiates the alteration in the composition of hydrotalcite layers in the presence of Fe^{3+} . These computed values align closely with those documented in previous studies (Millange *et al.*, 2000; Chebout *et al.*, 2010; León *et al.*, 2010; Kerchiche *et al.*, 2017).

The size of crystallite particles (D) was calculated using the Debye-Scherrer equation (Tichit *et al.*, 2006). For MgAl-h400, Fe/MgAl-h400, and Fe/MgAl-c400, the calculated D values are approximately 2.1 nm, 3.8 nm, and 3.5 nm, respectively. These results indicate an increase in particle size due to the presence of iron content, as demonstrated in the work by (Kerchich *et al.*, 2021). Furthermore, it is worth noting that the hydrothermal preparation method for Mg-Al confirmed an enhancement in crystallinity.

The SEM micrograph in Figure 4a reveals that the hydrothermal method yielded a synthetic MgAlh material exhibiting a plate-like structure characteristic of lamellar phases. Upon calcination, as depicted in the micrographs in Figure 4b and c, it becomes evident that the material exhibits enhanced crystallinity and an increase in particle size when compared to the hydrothermal method. Additionally, the surface of the hydrotalcite is conspicuously adorned with uniformly dispersed iron oxide nanoparticles, creating a relatively homogeneous appearance of γ -Fe2O3 (Barakat, 2012; Kerchich et al., 2021).



Fig.4. SEM images: (a) MgAl-h, (b) Fe/MgAl-h400 and (c) Fe/MgAl-c400

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Hence, an increase in the cristallinty signifies a reduction in the bandgap width, a concept supported by various studies (Burstein, 1954; Moss, 1954; Aliahmad & Nasiri Moghaddam, 2013). This relationship is corroborated by our assessment of the bandgap (Eg) in our samples using Tauc's equation (Tauc *et al.*, 1966), expressed as:

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$$(\alpha h \upsilon)^{m} = B (h \upsilon - Eg) (1)$$

In this equation, B represents a constant, Eg denotes the optical gap energy, α stands for the optical absorption coefficient, hv represents the energy of the photon, h is the Planck constant, and v signifies photon frequency. The value of Eg is determined by plotting $(\alpha h \upsilon)^m$ against (h υ) and extrapolating the linear portion of the curve that intersects the energy axis h υ . For direct gap materials, this corresponds to m = 2, whereas for indirect gap materials, it's m = 1/2.

Fe/MgAl-h400 exhibited an indirect band gap energy with a value of 1.89 eV, compared to Fe/MgAl-c400 exhibited a direct band gap energy with a values of 2.02 (Fig. 5). Thus, the band gap values of synthesis materials are slightly lower than that of the bulk Fe₂O₃ (2.1 eV). This widening of the prohibited band is an effect of the quantum confinement observed in many other synthesized semiconductor materials (Wen *et al.*, 2005; Bepari *et al.*, 2017). So, such narrow band gap is beneficial for the efficient utilization of visible light (Iron oxide exhibits a red color, which can absorb in the visible region). The narrower band gap of the material can harvest more photons to excite the e- from the valence band (VB) to the conduction band (CB) and thus, the photocatalytic activity is increased.



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Fig.5. Band gap energies of calcined Fe/MgAl-h400 and Fe/MgAl-c400

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310 Gap type effect on photocatalytic degradation by Fe/MgAl- photocatalyst

The photocatalytic activity of the synthesized Fe/MgAl-h400 material was evaluated for the degradation of MB dye in aqueous solution. The tests examined with the hydrothermal method under artificial irradiation using a tungsten lamp as visible light, under the same conditions which are applied for the evolution of the catalytic activity of the Fe/MgAl-c400 material prepared by the

coprecipitation method. Namely, the mass percentage of iron, the dye concentration, the amount of 315 photocatalyst, the temperature of calcinations and at a free pH (5.9). Using MB concentration of 50 316 mg/L and a catalyst amount of 0,5 g/L, it is can be easily observed that MB removal increased 317 efficiency with the reaction time (Fig.6). The yield obtained from the degradation of MB is as 318 expected due to the presence of iron oxide with a maghemite structure (γ -Fe₂O₃)(Kerchich *et al.*, 319 2021), that is a semiconductor generally known for its visible band gap energy (~ 2.1 eV). So, the 320 supported catalysts on an inert support as hydrotalcite structure increase the metal dispersion, and 321 the photo-activity depended particularly on the particle (Boudjemaa et al., 2009; Boudjemaa & 322 Trari, 2010). According to the data shown in Fig. 6, the highest reactivity is observed with with the 323 photocatalyst prepared by the Fe/MgAl-c400 coprecipitation. After 240 min of irradiation the 324 reaction efficiency is around of 67 and 46% for Fe/MgAl-c400 and Fe/MgAl-h400 photocatalysts, 325 respectively. 326

To compare the performance of two synthesis methods, hydrothermal and coprecipitation, in the 327 context of photodegradation, the results clearly indicate that MgAl-h400 exhibits lower 328 photodegradation. Despite their impressive fundamental properties as adsorbents, lamellar 329 structures are less frequently employed in photocatalytic reactions(Figueras, 2004; Figueras et al., 330 2006). Photocatalysis operates on the principle of degrading materials under the influence of light 331 with energy equal to or greater than its band gap ($h\nu \ge Eg$). This process involves catalyzing 332 chemical reactions through electronically excited species generated by photon absorption 333 (Khatamian et al., 2010; Dessie, 2017; Terna et al., 2021). 334

When the material is irradiated with a photon with energy greater than or equal to the band gap (Eg), it results in the generation of an electron-hole pair (e^{-}/h^{+}), leading to the formation of hydroxyl radicals (OH°) and superoxide anion radicals (O2°-), as well as the photogenerated hole (h^{+}) responsible for the degradation of organic molecules. Under irradiation, the h^{+} hole reacts with water to produce OH° radicals. It has been previously demonstrated by (Kerchich *et al.*, 2021), that OH° radicals and h⁺ holes are the reactive species involved in the photodegradation of MB (methylene blue) for Fe/MgAl-400 materials. A photocatalysis goes beyond the adsorption property of solids (adsorbents) to capture certain adsorbates on their surfaces, therefore does not is not just based on the property that solids have of fixing certain adsorbates on their surface. Hence, based on the gap results, we can conclude that the main difference in lies in the direct and indirect band gaps exhibited by the Fe/MgAl-c400 and Fe/MgAl-h400 semiconductors, respectively.

The fundamental distinction between direct and indirect band gaps lies in their crystal momentum 346 characteristics. In a direct band gap semiconductor, both electrons and holes possess identical 347 crystal momentum values in the conduction and valence bands, enabling direct photon emission by 348 an electron. Conversely, indirect band gap semiconductors lack this symmetry, making it 349 impossible for an electron to directly emit a photon since photons cannot carry crystal momentum. 350 For radiative recombination to take place in an indirect band gap material, it necessitates the 351 involvement of phonon absorption or emission, where the phonon's momentum compensates for the 352 difference between the electron and hole momenta. However, this process occurs at a considerably 353 slower pace because it relies on the intersection of three entities: an electron, a photon, and a 354 phonon. As a result, the recombination process is significantly more efficient in direct band gap 355 semiconductors compared to indirect band gap semiconductors. In the case of indirect band gap 356 semiconductors, the recombination process relies on phonon mediation, which results in a lower 357 photocatalytic efficiency compared to direct band gap semiconductors (Johari & Shenoy, 2012; 358 Conley et al., 2013; Gabal et al., 2023). In conclusion, a wider band gap increases the density of 359 electronic states in the material. This means that there are more sites available for chemical 360 reactions, which can facilitate the degradation of dyes. Excited electrons from the valence band can 361 more easily interact with other molecules and chemically react to degrade the dyes. 362





Fig.6. Effect of the synthesis method on the removal of MB (50ppm BM, 0.5g/l mass photocatalys,

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free pH (5,9))

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367 Conclusion

The research presented in this article contributes to the advancement of knowledge by focusing on 368 the development of novel photocatalysts based on iron oxide within the lamellar double hydroxide 369 370 structure, and their application in environmental contexts. Hydrotalcite-based materials have garnered significant interest in recent decades due to their simple synthesis processes and wide 371 range of applications. These materials are predominantly synthesized through co-precipitation and 372 hydrothermal methods, enabling the combination of M(II) and M(III) cations along with interlayer 373 anions within the hydrotalcite structure. The unique structural properties of hydrotalcite make it a 374 promising candidate for various applications, particularly in the field of photocatalysis. 375

376 This study successfully demonstrated the effectiveness of the hydrothermal method in synthesizing

a new photocatalyst, Fe/MgAl-h400, based on iron. The key findings of the research are as follows:

The Fe/MgAl-h400 photocatalyst was synthesized using the impregnation method and
 underwent comprehensive physico-chemical characterization, including XRF, XRD, SEM,
 ATG/DSC, RDs, and FTIR techniques.

- The infrared spectra exhibited typical characteristics associated with lamellar double
 hydroxides, consistent with previous studies on hydrotalcites, and the XRD results clearly
 indicated the presence of iron oxide.
- The Fe/MgAl-h400 photocatalyst demonstrated improved crystallinity and remarkable photoactivity under visible light. After 240 minutes of irradiation, the Fe/MgAl-h400 photocatalyst, with a particle size of approximately 3.8 nm and an indirect bandgap energy of 1.89 eV, exhibited a notable photodegradation rate of approximately 46%.
- The Fe/MgAl-h400 semiconductor, with an indirect bandgap, undergoes a recombination process that depends on phonon mediation, leading to lower photocatalytic efficiency compared to direct bandgap semiconductors. However, a wider bandgap in semiconductor materials provides notable advantages for dye degradation, including enhanced light absorption, increased reactivity with dyes, greater stability, and improved durability.
- Therefore, enhancing the qualities and modifying the basic properties of mixed oxides
 derived from hydrotalcite precursors is highly beneficial. These modifications facilitate
 reactions that require varying basic strength and contribute to more efficient and controlled
 degradation of dyes, which is crucial in fields such as wastewater treatment, air purification,
 and the removal of unwanted dyes.

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