

# Preparation of mesoporous TiO<sub>2</sub> nanomaterials doped with rare earth ions (La<sup>3+</sup>, Sm<sup>3+</sup>, Nd<sup>3+</sup>, Gd<sup>3+</sup>) and its application in the photodegradation of unsymmetrical dimethylhydrazine wastewater

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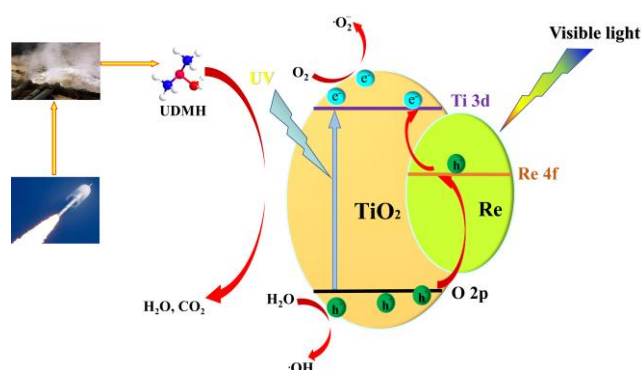
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## Graphical abstract



## Abstract

Methyldimethylhydrazine and its by-products cause serious damage to human health. In this paper, Re<sup>3+</sup>-doped mesoporous TiO<sub>2</sub> were synthesized for the photodegradation of UDMH. The structure, morphology and optical properties of photocatalyst were extensively characterized. The photocatalytic degradation was significantly enhanced, and 2%Nd-TiO<sub>2</sub> showed the most excellent photocatalytic degradation performance with 93.3% degradation rate of UDMH. NDMA and FDMH were gradually and completely degraded. O<sub>2</sub><sup>·-</sup> and ·OH played an important role in the photocatalytic degradation of UDMH. The possible degradation pathways of UDMH analyzed by GC-MS. The work provided an effective method for the degradation of UDMH wastewater.

**Keywords:** mesoporous TiO<sub>2</sub>, degradation, unsymmetrical dimethylhydrazine

## 1. Introduction

UDMH has been widely used as a high-energy liquid rocket fuel for missile weapons and aerospace applications because of its high specific impulse, low cost and easy storage. (Liu *et al.*, 2012) In the process of space launch, engine testing, fuel storage and transportation, a large amount of UDMH exhaust gas and waste liquid will be generated, which is prone to cancer and genetic mutation. (Carlsen *et al.*, 2009) The transformation

products of UDMH such as nitrosodimethylamine (NDMA) and metahydrazone (FDMH) are highly carcinogenic [Sgroi *et al.*, 2018], Therefore, it is of great practical significance to study the degradation and mineralization methods of methyldimethylhydrazine. Traditional treatment methods of methyldimethylhydrazine mainly include physical treatment, chemical treatment and biological treatment. Traditional physical treatment methods such as activated carbon adsorption and ion exchange can remove UDMH from wastewater smoothly, but activated carbon and ion exchange resin cannot be regenerated and recycled, which is easy to cause secondary pollution. (FleMing *et al.*, 1996) Chemical treatment methods such as H<sub>2</sub>O<sub>2</sub>/O<sub>3</sub>/UV degradation of UDMH in wastewater (Ismagilov *et al.*, 2002). Such methods are costly, energy intensive, and require MnO<sub>2</sub> and other substances to degrade the excess H<sub>2</sub>O<sub>2</sub> in the experiment, and O<sub>3</sub> will corrode the experimental equipment. Biological treatment methods mainly use aquatic plant absorption method and microbial degradation treatment, the treatment cycle is long and slow, and it is difficult to deal with large-scale high concentration of UDMH wastewater, and the above traditional methods all have certain defects.

Photocatalysis is a sustainable oxidation technology that directly uses solar energy and is an effective solution for degrading pollutants in water bodies (Rueda-Marquez *et al.*, 2020). TiO<sub>2</sub> photocatalysts are green and economical, highly active and non-selective in the degradation of water pollutants (Rashid *et al.*, 2023) which can mineralize potential pollutants into non-polluting substances, and is a hot spot for photocatalyst research (Gopinath *et al.*, 2020; Rahman *et al.*, 2023; Malik *et al.*, n.d.). However, the currently prepared TiO<sub>2</sub> photocatalytic materials generally have a wide band gap (3.0~3.2eV) responding to UV light only, with only 4%-5% of the available solar energy, limited use in visible light (400 nm < λ < 700 nm), and high photogenerated electron and hole complexation rates with low catalytic efficiency (Byrne *et al.*, 2018). It is found that doping can extend the spectral response range and reduce the complexation rate of electron-hole pairs,

thus improving the photocatalytic performance (Paradisi *et al.*, 2023; Liang *et al.*, 2023; Escamilla-Mejia *et al.*, 2023), Rare earth metals produce a high number of energy levels because they have unfilled 4f shell layers or empty 5d orbitals, while the 4f, 5d, and 6s electrons are close in energy (Lee *et al.*, 2013), exhibits unique properties and functions. Studies have shown (Wang *et al.*, 2011) that the doping modification of rare earth elements can effectively capture photogenerated electrons and inhibit the recombination of photogenerated electrons and holes, which can improve the photocurrent conversion efficiency of photocatalysts in the wavelength range of 300~400 nm, redshift the absorption sidebands, improve the utilization of visible light, and greatly improve the photocatalytic efficiency, and rare earth metal doping has now been widely used in the doping modification of photocatalysis (Zhao and Liu, 2008; Saif *et al.*, 2014; El-Bahy *et al.*, 2009), such as Eu-doped TiO<sub>2</sub> (Juan *et al.*, 2022), La-doped TiO<sub>2</sub> (Huang *et al.*, 2017), Nd-doped TiO<sub>2</sub> (Wang *et al.*, 2019), Er-doped TiO<sub>2</sub> (Chen *et al.*, 2022).

The preparation of TiO<sub>2</sub> nanomaterials with mesoporous structure is an effective way to enhance photocatalytic activity. Its large specific surface area and well-developed pore structure can facilitate the adsorption of reactants and the absorption of light energy, and it can generate more active reaction centers, thus improving photocatalytic activity (Hou *et al.*, 2020). There are various synthetic methods such as hydrothermal method, template method and sol-gel method to prepare mesoporous TiO<sub>2</sub> nanomaterials, however, it is still challenging to prepare mesoporous TiO<sub>2</sub> nanomaterials with high specific surface area and high crystallinity because the grain growth may lead to collapse of mesoporous structure.

In this study, Re<sup>3+</sup> (La<sup>3+</sup>, Sm<sup>3+</sup>, Nd<sup>3+</sup>, Gd<sup>3+</sup>)-doped mesoporous titanium dioxide nanopowders were synthesized using a low energy consumption and low pollution hydrothermal method with simple organic reagents and inorganic acids, and their physicochemical properties were characterized by using various characterization tools, and the effects of rare earth ion doping on the crystal structure, surface chemical state, morphology, optical properties and the effects of rare earth ion doping on the crystal structure, surface chemical state, morphology, optical properties and activity of photocatalytic degradation of meta-dimethylhydrazine were investigated in detail. The intermediates in the degradation process were analyzed by GC-MS, and the reactive radicals were analyzed by radical capture experiments and EPR to deduce the possible mechanism of photocatalysis.

## 2. Experiment

### 2.1. Materials and reagents

UDMH, (98%), NDMA(>98%), O<sub>2</sub>si, FDMH(>98%), Aladdin Chemicals, Tetrabutyl titanate, Aladdin Chemicals, Inc. (>99%), HCL (AR), Xi 'an Mipu Chemical Reagent Co., LTD., HNO<sub>3</sub> (AR), Xi 'an Mipu Chemical Reagent Co., LTD., NaOH (AR), Xi 'an Mipu Chemical Reagent Co., LTD., anhydrous

ethanol (AR) Tianjin Hengxing Chemical Reagent Co., LTD., La<sub>2</sub>O<sub>3</sub>, Sm<sub>2</sub>O<sub>3</sub>, Nd<sub>2</sub>O<sub>3</sub>, Gd<sub>2</sub>O<sub>3</sub>, Aladdin Chemical Company 99.99%), isopropyl alcohol, p-benzoquinone, Disodium EDTA, Carbon Tetrachloride, Aladdin Chemicals Co., LTD. (>99.99%), deionized water, laboratory homemade. all reagents are analytical grade reagents, no further treatment is required when used.

### 2.2. Preparation of photocatalyst

As shown in Figure 1, the rare-earth modified mesoporous TiO<sub>2</sub> nanoparticles were synthesized by a one-step hydrothermal method. 15ml of tetrabutyl titanate was added dropwise to 20 ml of anhydrous ethanol with vigorous stirring for 60 min to obtain a light yellow transparent solution A. A certain concentration of HCL solution was added dropwise to solution A to obtain mixed solution B. A certain amount of lanthanide oxide was weighed and dissolved in HNO<sub>3</sub> to prepare 100ml of 50mM Re(NO)<sub>3</sub> solution was set aside, add different volumes of 50mM Re(NO)<sub>3</sub> solution drop by drop in solution B, stirred for 30min to ensure that the material ratio of rare earth Re<sup>3+</sup> to TiO<sub>2</sub> was 0.5%, 1%, 2% and 3% respectively and mixed well, slowly added NaOH solution drop by drop to the above suspension, adjusted the pH to about 7 and stirred for 1h, in turn the suspension was transferred to a 100ml autoclave, put into an oven at 150°C/24h, and the supernatant was decanted after it cooled naturally to room temperature, centrifuged, rinsed with deionized water and anhydrous ethanol, filtered and dried, and ground into powder to obtain the sample TiO<sub>2</sub>, 0.5%Re<sup>3+</sup>-TiO<sub>2</sub>, 1%Re<sup>3+</sup>-TiO<sub>2</sub>, 2%Re<sup>3+</sup>-TiO<sub>2</sub>, 3%Re<sup>3+</sup>-TiO<sub>2</sub> are simply noted as TiO<sub>2</sub>, 0.5%Re-TiO<sub>2</sub>, 1%Re-TiO<sub>2</sub>, 2%Re-TiO<sub>2</sub>, 3%Re-TiO<sub>2</sub>.

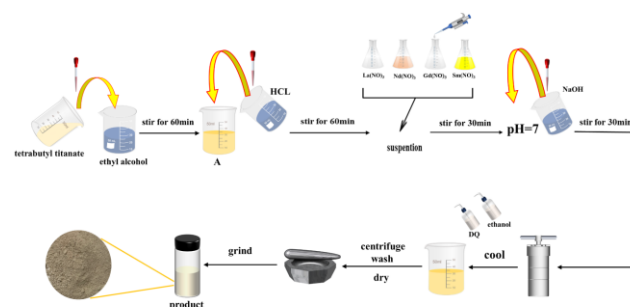


Figure 1. Material preparation process diagram

### 2.3. Characterization

The physical composition and crystal image of the samples were examined by X-ray diffractometer (XRD, bruker D8 advanced, Germany) with the following operating parameters: Cu target, K $\alpha$  rays,  $\lambda=0.15401$  nm, target voltage 40KV, scanning range 5°~90°, scanning speed 5°/min, FT-IR spectroscopy (FT-IR, bruker VERTEX 70, Germany) to measure the FT-IR spectra of the samples in the range of 450 cm<sup>-1</sup>~4000 cm<sup>-1</sup>, KBr was used as diluent, and the elemental composition of the material surface was determined by X-ray photoelectron spectrometer (XPS, AXIS ULTRADLD, UK) and analyzed for its chemical valence state, combined with the ability to use the C 1s peak at 284.8 eV for calibration. The surface morphology of the material was observed by a cold emission scanning

electron microscope (FESEM, Gemini, Germany) equipped with EDS at a high speed voltage of 15 kV, and the elemental composition and relative content of the material were determined by an energy spectrometer (EDS). The microscopic morphology and crystal structure of the materials were further observed by a high-resolution transmission electron microscope (HRTEM, JEM-2100F, Japan), and the crystalline spacing was analyzed, and the absorption properties of the UV-Vis diffuse reflectance spectra were tested by a UV-Vis spectrometer (UV-Vis DRS, Ocean Optics SB650, USA). PL steady-state spectra with excitation wavelength of 325 nm, Raman spectra were obtained by nanoraman spectrometer (Naman, HORIBA JY HR-800, France) under 532 nm excitation, the specific surface area of the material was measured by specific surface area and porosity analyzer (JW-BK300, China), and the pore size distribution was calculated by BJH method, and the pore size distribution was calculated by electron paramagnetic resonance spectrometer (EPR, Bruker A300, Germany) to collect electron paramagnetic resonance spectra, detect and analyze free radicals, photoelectrochemical measurements were performed at an electrochemical workstation (Priston P4000, USA) by the chronocurrent method, FDMH, NDMH by high performance liquid chromatography (HPLC, Ultimate 3000, USA)

#### 2.4. Analysis method

The photocatalytic activity of the prepared samples was evaluated by decomposing the dye under a 350W xenon lamp, using a CEL-LAB500E 350W xenon lamp with AM 1.5 filter to simulate sunlight as the catalytic light source, and the light intensity was set to 100 mW/cm<sup>2</sup>. The catalyst was stirred for 30 min to ensure the adsorption-desorption equilibrium, and the suspension was centrifuged to measure the absorbance A value of the upper clear layer.

The degradation efficiency R, mineralization efficiency and reaction kinetics of UDMH were used to evaluate the photocatalytic performance of the material with the following equations:

$$R = (C_0 - C_t) / C_0 \quad (1)$$

Where: C<sub>0</sub> is the initial concentration of UDMH, mg/L; C<sub>t</sub> is the concentration of UDMH in the wastewater at the time of photocatalytic reaction t, mg/L. In addition, the general photocatalyst degradation of organic pollutants is a pseudo primary kinetic reaction with the following equation:

$$\ln \frac{C_0}{C_t} = kt \quad (2)$$

Where: k is the first-order kinetic reaction rate constant, h<sup>-1</sup> or min<sup>-1</sup>; t is the time of photocatalytic degradation of UDMH

$$\text{Mineralization rate (100\%)} = \frac{\text{TOC}_0 - \text{TOC}_t}{\text{TOC}_0} \quad (3)$$

Where: TOC<sub>0</sub> and TOC<sub>t</sub> represent the concentration of total organic carbon at the initial time and reaction time t.

The method used for the determination of UDMH in the experiment was the spectrophotometric method of sodium aminoferricyanide in GB/T 14376-1993 (Zhang *et al.*, 2023), With UV-vis spectrometer (721, Shanghai Instrument Analysis Factory, China), UDMH reacted with sodium aminocyanide to form a red complex with a maximum absorption wavelength of 500 nm, and the concentration of UDMH was determined by the principle that the color depth of the red complex is proportional to the content of UDMH (Zhou *et al.*, 2020). Further assessment of mineralization of methyldimethylhydrazine wastewater by means of a total organic carbon analyzer (vario TOC select, Elementar Analysensysteme GmbH, Germany).

#### 2.5. Photoelectrochemical performance analysis

The photoelectrochemical performance of the photocatalyst was tested by three-electrode system and electrochemical workstation, the photocatalyst was loaded on ITO conductive glass to prepare working electrode, Pt sheet electrode as counter electrode, Ag/AgCl electrode as reference electrode, 0.5 mol/L Na<sub>2</sub>SO<sub>4</sub> as electrolyte, and the photoelectrochemical performance was tested by chrono-current method, and the current density was measured under light and dark conditions in turn, and the separation efficiency was judged by The photocurrent changes of photoelectrons and holes in the external circuit after excitation were used to determine the degree of fast response of the material to light and the separation efficiency of photoelectron-hole pairs, the higher the photocurrent, the higher the separation efficiency.

#### 2.6. Free radical capture experiment

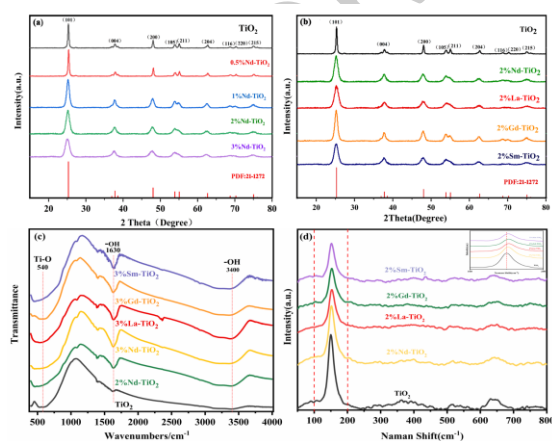
According to the basic principle of photocatalysis, the degradation process of organic matter mainly consists of direct oxidation of organic pollutants by photogenerated pairs, or the reaction with surface adsorbed oxygen (O<sub>2</sub>) and H<sub>2</sub>O/OH<sup>-</sup> to generate superoxide radicals(O<sub>2</sub><sup>-</sup>) and hydroxyl radical (OH). In order to determine the mechanism of photocatalytic degradation of methyldimethylhydrazine, different radical trapping agents were added to the photocatalytic degradation of methyldimethylhydrazine under the same experimental conditions, and the role of each radical in the photocatalytic degradation process was evaluated by verifying the type of reacting radicals, and this paper mainly used isopropyl alcohol, p-benzoquinone, disodium ethylenediaminetetraacetic acid (Na<sub>2</sub>EDTA), and carbon tetrachloride as capture agent of OH·O<sub>2</sub><sup>-</sup>h<sup>+</sup> and e<sup>-</sup>, Set the addition concentration at 1 mM and analyze the effect of active species produced by different photocatalysts on the degradation of UDMH wastewater, using the UDMH wastewater degradation system with only the catalyst added as a blank reference, no trapping agent was added and other conditions were kept consistent (Alam *et al.*, 2018).

### 3. Results and discussions

#### 3.1. Structural analysis of photocatalysts

The crystal structure of the samples was studied by X-ray diffractometer (XRD). Figure 2 shows that the XRD spectra

of pure  $\text{TiO}_2$  and rare-earth  $\text{Re}^{3+}$ -doped  $\text{TiO}_2$  are in basic agreement with the standard card of anatase phase  $\text{TiO}_2$  (PDF:21-1272), the characteristic diffraction peaks of each crystal plane family in the spectra are sharp, and the material has good crystallization properties. The  $2\theta$  characteristic diffraction peaks at  $25.3^\circ$ ,  $37.5^\circ$ ,  $48.2^\circ$ ,  $55.1^\circ$ ,  $62.9^\circ$  correspond to the (101), (004), (200), (105), (204) crystallographic planes of anatase titanium dioxide crystals (Fan *et al.*, 2014). The diffraction peaks of rare earth  $\text{Re}^{3+}$ -doping are not significantly different from those of pure  $\text{TiO}_2$ , and the diffraction peaks of rare earth oxides do not appear after doping, indicating that the rare earth elements are uniformly dispersed in the  $\text{TiO}_2$  phase after doping, and there is no independent phase (Hassan *et al.*, 2012). Considering the radii of  $\text{Nd}^{3+}$  (0.100 nm),  $\text{La}^{3+}$  (0.106 nm),  $\text{Sm}^{3+}$  (0.096 nm),  $\text{Gd}^{3+}$  (0.094 nm), which are much larger than the radius of  $\text{Ti}^{4+}$  (0.068 nm), it is difficult for  $\text{Re}^{3+}$  to directly enter the lattice of  $\text{TiO}_2$  to replace  $\text{Ti}^{4+}$  to form a stable solid solution, but it is uniformly dispersed in the form of clusters in  $\text{TiO}_2$  nanoparticles or in the interstices of the iron-oxygen octahedra of titanium dioxide (Nguyen-Phan *et al.*, 2009). In Figure 2 (b), although the different  $\text{Re}^{3+}$  doping process did not change the crystalline phase of  $\text{TiO}_2$ , the diffraction peak half-peak width gradually increased and intensity gradually decreased compared with the undoped sample, and (101) the crystalline surface shifted to a low angle, and in Figure 2 (a), when  $\text{Nd}^{3+}$  was doped, the doping ratio gradually increased, and the trend of peak broadening and dwarfing became more obvious, the different  $\text{Re}^{3+}$  doping inhibits the growth of titanium dioxide grains and causes grain refinement, and within a certain doping ratio, the trend of grain refinement is more obvious as the amount of  $\text{Re}^{3+}$  doping increases. The grain refinement can increase the specific surface area and enhance the oxidation ability of the photocatalyst through the quantum size effect, which will be beneficial to the non-homogeneous photocatalytic reaction.



**Figure 2.** (a) XRD spectra of pure  $\text{TiO}_2$  and  $\text{TiO}_2$  doped with different concentrations of  $\text{Nd}^{3+}$ . (b) XRD spectra of pure  $\text{TiO}_2$  and 2%  $\text{Re-TiO}_2$  ( $\text{Nd}^{3+}$ ,  $\text{La}^{3+}$ ,  $\text{Gd}^{3+}$ ,  $\text{Sm}^{3+}$ ) (c) FT-IR spectra of pure  $\text{TiO}_2$ , 2%  $\text{Nd-TiO}_2$  and 3%  $\text{Re-TiO}_2$  ( $\text{Nd}^{3+}$ ,  $\text{La}^{3+}$ ,  $\text{Gd}^{3+}$ ,  $\text{Sm}^{3+}$ ) (d) Raman spectra of pure  $\text{TiO}_2$  and 2%  $\text{Re-TiO}_2$  ( $\text{Nd}^{3+}$ ,  $\text{La}^{3+}$ ,  $\text{Gd}^{3+}$ ,  $\text{Sm}^{3+}$ )

The information of the chemical structure of the samples was further investigated by FT-IR test, and Figure 2 (c) shows that all samples showed three characteristic

absorption peaks at  $540\text{ cm}^{-1}$ ,  $1630\text{ cm}^{-1}$ , and  $3400\text{ cm}^{-1}$ , which were attributed to the Ti-O stretching vibration in anatase  $\text{TiO}_2$ , the bending vibration of physisorbed water molecules -OH, and the stretching of surface -OH from water dissociation vibration, compared with pure  $\text{TiO}_2$ , the characteristic peak at  $1630\text{ cm}^{-1}$  is significantly enhanced after doping, and the peak of 3%  $\text{Nd-TiO}_2$  is stronger than that of 2%  $\text{Nd-TiO}_2$ , probably because after  $\text{Re}^{3+}$  doping,  $\text{Ti}^{4+}$  replaces  $\text{Re}^{3+}$  at the rare-earth oxide lattice interface and forms Ti-O-Re bond leading to lattice distortion, which increases the number of surface-OH, and at the same time the Ti-O characteristic peak is red-shifted. The  $3440\text{ cm}^{-1}$  characteristic absorption peak indicates that the photogenerated  $h^+$  generated by  $\text{TiO}_2$  after photoexcitation reacts with  $\text{H}_2\text{O}$  or  $\text{OH}^-$  to form  $\cdot\text{OH}$ . The enhancement of the peak at  $3440\text{ cm}^{-1}$  after doping indicates that the number of  $\cdot\text{OH}$  increases after doping, which is beneficial to the photocatalytic reaction activity, and this change may be due to the increase of specific surface area after  $\text{Re}^{3+}$  doping. The new characteristic peaks did not appear after doping, and only some of the peaks were broadened and distorted, which is a change in the interaction forces between the groups only after doping, which corroborates with the XRD results.

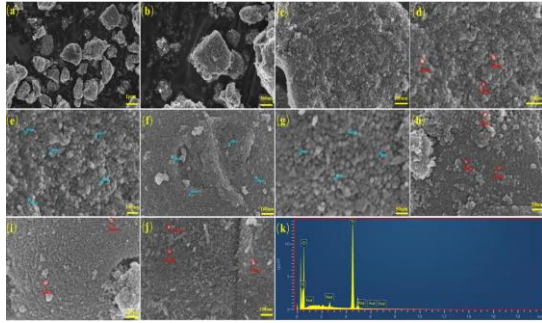
The crystal structure and surface defects of the samples were further investigated by Raman tests.  $144\text{ cm}^{-1}(E_g)$ ,  $197\text{ cm}^{-1}(E_g)$ ,  $639\text{ cm}^{-1}(E_g)$ ,  $513\text{ cm}^{-1}(A_{1g})$ , and  $396\text{ cm}^{-1}(B_{1g})$  Raman vibrational peaks revealed the synthesis of anatase phase  $\text{TiO}_2$  (Kibasomba *et al.*, 2018). After  $\text{Re}^{3+}$  ( $\text{Nd}^{3+}$ ,  $\text{La}^{3+}$ ,  $\text{Gd}^{3+}$ ,  $\text{Sm}^{3+}$ ) doping, The strongest peak at  $144\text{ cm}^{-1}(E_g)$  is obviously weakened, indicating that the particle size of  $\text{TiO}_2$  decreases with  $\text{Re}^{3+}$  doping, and the anatase phase  $\text{TiO}_2$  is a tetragonal crystal system, and the change of  $E_g$  vibrational peak in the tetragonal crystal system can directly prove the generation of oxygen vacancies (Salari *et al.*, 2011). The inset in the upper right corner of Figure 2 (d) shows the strongest peak of the  $E_g$  pattern for all samples, which is shifted toward the high wave number, indicating that more oxygen vacancies are generated in the doped titanium dioxide nanolattice structure, and the oxygen vacancies can act as electron capture traps to inhibit carrier complexation. In addition, the oxygen vacancies can also form transition energy levels between the conduction band and valence band, reducing the band gap and thus enhancing the visible light response, thus enhancing the photocatalytic activity (Jiang *et al.*, 2012).

### 3.2. Morphological analysis of photocatalysts

From Figure 3, it can be seen that the prepared photocatalysts are porous materials composed of inhomogeneous nanoparticles, and the phenomenon of particle agglomeration occurred in all samples. The prepared pure  $\text{TiO}_2$  is a group of irregular block structures with more fragments and cracks, with agglomerated morphology, and the particle size is about  $30\sim 40\text{ nm}$ . The doping of rare earth  $\text{Re}^{3+}$  ( $\text{Nd}^{3+}$ ,  $\text{Sm}^{3+}$ ,  $\text{Gd}^{3+}$ ,  $\text{La}^{3+}$ ) has an obvious inhibiting effect on the growth of  $\text{TiO}_2$ . The grain size of 1% $\text{Nd-TiO}_2$  is about  $15\sim 25\text{ nm}$ , 2% $\text{Nd-TiO}_2$  is about  $10\sim 20\text{ nm}$ , 3% $\text{Nd-TiO}_2$  is about  $10\sim 15\text{ nm}$ , with the increase



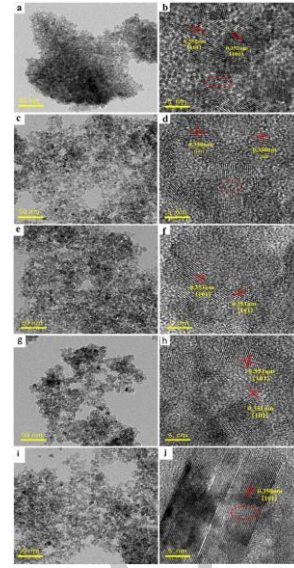
of Nd<sup>3+</sup> doping, the trend of grain refinement is more obvious, which is consistent with the xrd results and can be obtained from the increase of specific surface area in Table 1 corroborate (Parveen *et al.*, 2017). In order to determine the actual doping amount of Re<sup>3+</sup> in Nd-TiO<sub>2</sub>, an energy spectrometer (EDS) was used to test and analyze the content of 2% Nd-TiO<sub>2</sub>. The peaks identified in the EDS plot corresponded to Ti, O, and Nd elements, and the mass ratio of Nd to TiO<sub>2</sub> was calculated to be 2.03%, which was basically consistent with the expected doping amount, and the EDS results indicated the successful synthesis of 2%Nd-TiO<sub>2</sub> photocatalyst was successfully synthesized.



**Figure 3.** FESEM plots of (a) (b) (c) (d) pure TiO<sub>2</sub>, (e) 1%Nd-TiO<sub>2</sub>, (f) 2%Nd-TiO<sub>2</sub>, (g) 3%Nd-TiO<sub>2</sub> (h) 3% Sm-TiO<sub>2</sub>, (i) 2%Gd-TiO<sub>2</sub>, (j) 2%La-TiO<sub>2</sub> (k) EDS plots of 2%Nd-TiO<sub>2</sub>

HRTEM images can provide finer crystal internal results and analyze the crystalline surface spacing and defects. From Figure 4, The prepared samples have a particle size of about 10~30 nm, and the morphology is an aggregated state of nanoparticles, and the nanoparticles stack up to form porous channels similar to wormholes, which are randomly distributed between the nanoparticles and form a three-dimensional structure, which can increase the specific surface area of the catalyst, and then provide more catalytic reaction active sites, the prepared pure TiO<sub>2</sub> and Re<sup>3+</sup>-doped TiO<sub>2</sub> lattice stripes are clear, indicating good crystallinity, and the lattice stripe spacing of 0.352 nm, 0.351 nm, and 0.350 nm correspond to the (101) crystallographic plane of anatase TiO<sub>2</sub>, and because the radius of Re<sup>3+</sup> is larger than that of Ti<sup>4+</sup>, the lattice distortion caused by the doping of Re<sup>3+</sup>, locally (within the dashed box ) lattice stripe produces a slight expansion, and the generated impurity defects or oxygen vacancies can be used as traps for photogenerated electrons or holes to inhibit

photogenerated electron-hole complexation and thus enhance photocatalytic activity.



**Figure 4.** HRTEM images of (a), (b) pure TiO<sub>2</sub>, (c), (d) 2%La-TiO<sub>2</sub>, (e), (f) 2%Gd-TiO<sub>2</sub>, (g), (h) 2%Nd-TiO<sub>2</sub>, (i), (j) 2%Sm-TiO<sub>2</sub>

The physical and chemical conditions of the catalyst surface directly determine the catalytic reaction active sites and adsorption, which have important effects on the photocatalytic reaction. The specific surface area and pore size distribution of the samples were analyzed by BET method, and the nitrogen adsorption and desorption curves and pore size distribution curves are shown in Figure 5. The specific surface area and pore size (DP) data of the samples are included in Table 1. At lower relative pressures ( $P/P_0 < 0.4$ ), the adsorption of the material is high, indicating the presence of mesoporous structure, and a type IV curve appears at relative pressures  $P/P_0 = 0.4 \sim 0.8$  with H<sub>4</sub> and H<sub>2</sub> hysteresis loops, indicating that the material is a typical mesoporous material (Thommes *et al.*, 2015), containing "ink bottle" interstitial pores. This is consistent with the results of the porous channels in the SEM photographs, where monolayer adsorption occurs mainly on the mesoporous pore walls at low relative pressures, with minimal micropore adsorption; at moderate relative pressures, adsorption increases rapidly and capillary coalescence occurs; and at high relative pressures, adsorption limitation occurs.

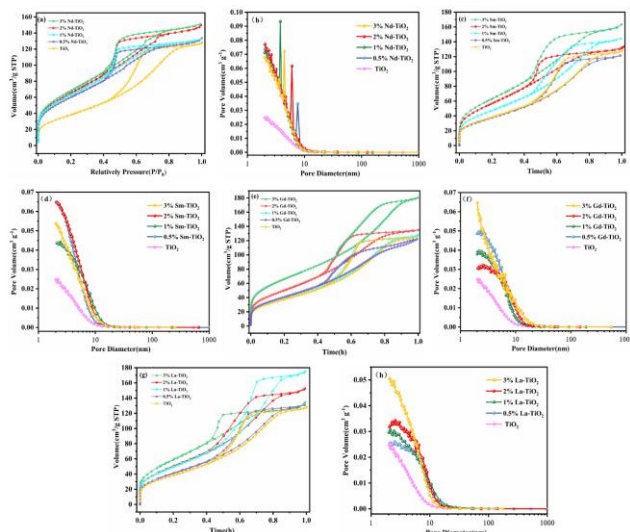
**Table 1.** BET specific surface area (SBET), pore volume and average pore size of the prepared samples

Sample	S <sub>BET</sub> (m <sup>2</sup> /g)	Pore volume (cm <sup>3</sup> /g)	Pore size (nm)	Sample	S <sub>BET</sub> (m <sup>2</sup> /g)	Pore volume (cm <sup>3</sup> /g)	Pore size (nm)
Pure TiO <sub>2</sub>	130.5	0.196	5.455	3%Gd-TiO <sub>2</sub>	172.4	0.237	4.323
3%Sm-TiO <sub>2</sub>	184.8	0.251	3.971	2%Gd-TiO <sub>2</sub>	155.4	0.215	4.378
2%Sm-TiO <sub>2</sub>	174.3	0.223	4.037	1%Gd-TiO <sub>2</sub>	142.2	0.203	4.613
1%Sm-TiO <sub>2</sub>	165.2	0.201	4.362	0.5%Gd-TiO <sub>2</sub>	133.4	0.196	4.939
0.5%Sm-TiO <sub>2</sub>	149.6	0.198	5.016	3%La-TiO <sub>2</sub>	163.7	0.218	5.155
3%Nd-TiO <sub>2</sub>	187.7	0.230	3.494	2%La-TiO <sub>2</sub>	151.1	0.205	5.246
2%Nd-TiO <sub>2</sub>	180.3	0.217	3.667	1%La-TiO <sub>2</sub>	145.5	0.199	5.363
1%Nd-TiO <sub>2</sub>	160.5	0.204	3.697	0.5%La-TiO <sub>2</sub>	139.3	0.197	5.433
0.5%Nd-TiO <sub>2</sub>	153.8	0.201	3.786				

The pore size distribution diagram shows that the pore size distribution of the prepared samples are all located in

the narrow range of 2~10 nm, which indicates that the intermediate pore distribution of the samples is

concentrated and uniformly ordered. The specific surface area and pore volume of 3%Sm-TiO<sub>2</sub>, 3%Nd-TiO<sub>2</sub> and 2%Nd-TiO<sub>2</sub> are larger, which can expose more active sites in the photocatalytic process and thus enhance the photocatalytic performance.

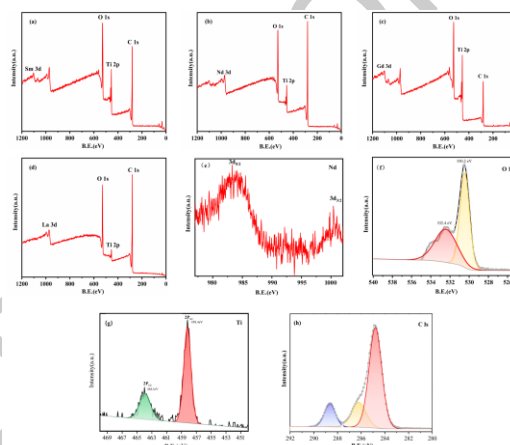


**Figure 5.** (a) N<sub>2</sub> adsorption-desorption isotherms of pure TiO<sub>2</sub> and Nd-TiO<sub>2</sub>, (b) pore size distribution curves of pure TiO<sub>2</sub> and Nd-TiO<sub>2</sub>, (c) N<sub>2</sub> adsorption-desorption isotherms of pure TiO<sub>2</sub> and Sm-TiO<sub>2</sub>, (d) pore size distribution curves of pure TiO<sub>2</sub> and Sm-TiO<sub>2</sub>, (e) N<sub>2</sub> adsorption-desorption isotherms of pure TiO<sub>2</sub> and Gd-TiO<sub>2</sub>, (f) pore size distribution curves of pure TiO<sub>2</sub> and Gd-TiO<sub>2</sub>, (g) N<sub>2</sub> adsorption-desorption isotherms of pure TiO<sub>2</sub> and La-TiO<sub>2</sub>, (h) pore size distribution curves of pure TiO<sub>2</sub> and La-TiO<sub>2</sub>.

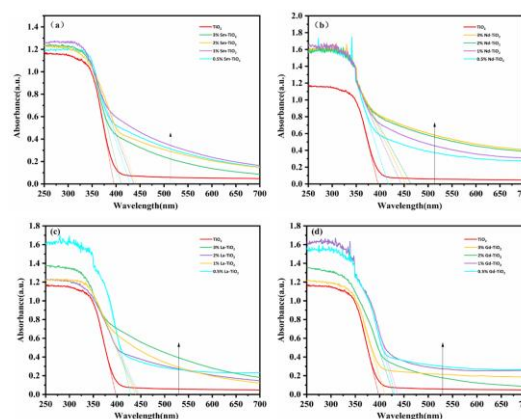
### 3.3. Analysis of chemical composition of photocatalysts

The elemental nature of the surface layer of the photocatalyst was analyzed by XPS to further determine the presence state of each element, and the total spectra in Figure 6 (a), (b), (c) and (d) show that the materials all contain Ti, Re (Sm, Nd, Gd, La), O and C. Figure 7 (e), (f), (g) and (h) are the XPS spectra of Nd 3d, O 1s, Ti in 2%Nd-TiO<sub>2</sub> 2p and C 1s XPS spectra, the characteristic peaks of Nd are located at 983.4 eV and 1003.6 eV, respectively, which represent the bond energies of Nd 3d<sub>5/2</sub> and Nd 3d<sub>3/2</sub> (Yuvakkumar and Hong, 2015). The asymmetric peaks of O 1s appear in two separate peaks at binding energies 530.2eV and 532.4eV, respectively, indicating the existence of two different types of oxygen species on the surface after doping modification, where the strong peak at 530.2eV is the Ti-O<sub>lattice</sub> chemical bond formed between titanium and oxygen elements in the lattice, corresponding to the lattice oxygen species, and the weak peak at 532.4eV is the Ti-O<sub>surf</sub> chemical bond formed between titanium and oxygen elements on the surface of the catalyst, corresponding to the lattice oxygen species. elements on the catalyst surface, and the corresponding oxygen species are surface hydroxyl oxygen and oxygen vacancies (Xing *et al.*, 2013). The XPS peaks of Ti 2p at binding energies 458.4 eV and 464.1 eV belong to Ti 2p<sub>3/2</sub> and Ti 2p<sub>1/2</sub>, respectively, and the characteristic peaks of Ti 2p<sub>3/2</sub> and Ti 2p<sub>1/2</sub> in the Nd-modified catalysts are shifted towards lower binding energies compared to those of pure TiO<sub>2</sub> at binding energies 458.5 eV and 464.5 eV (Nair *et al.*, 2023), which indicates that the 2% Nd-TiO<sub>2</sub>,

in addition to a large amount of Ti<sup>4+</sup>, there is also a small amount of Ti<sup>3+</sup>, and the radius of Nd<sup>3+</sup> (0.100 nm) is much larger than Ti<sup>4+</sup> (0.068 nm) and smaller than O (0.140 nm), so it is difficult for Nd<sup>3+</sup> to directly enter TiO<sub>2</sub> to replace Ti<sup>4+</sup>, but Ti<sup>4+</sup> may replace Nd<sup>3+</sup> at the rare-earth oxide lattice interface, which can form Ti-O-Nd bond, because Ti<sup>4+</sup> is more electronegative than Nd<sup>3+</sup>, it is easy to get electrons and make its characteristic peak move to low binding energy, and part of Ti<sup>4+</sup> is reduced to Ti<sup>3+</sup> (Nguyen-Phan *et al.*, 2009). The formation of Ti<sup>3+</sup> and the oxygen vacancies on the catalyst surface facilitate the photocatalytic activity there is a clear splitting of the C 1s peak, with the strong peak at 284.8 eV being the XPS-calibrated C contamination, while the other two weaker binding energy peaks are formed by the carbon-containing material in the air absorbed by the sample.



**Figure 6.** (a) Sm<sup>3+</sup>-doped TiO<sub>2</sub> photocatalyst XPS full spectrum; (b) Nd<sup>3+</sup>-doped TiO<sub>2</sub> photocatalyst XPS full spectrum; (c) Gd<sup>3+</sup>-doped TiO<sub>2</sub> photocatalyst XPS full spectrum; (d) XPS full spectrum of La<sup>3+</sup>-doped TiO<sub>2</sub> photocatalyst; (e) high-resolution X-ray photoelectron spectra of Nd 3d (f) O 1s (g) Ti 2p (h) C 1s;



**Figure 7.** (a) Pure TiO<sub>2</sub> and different concentrations of Sm<sup>3+</sup>-doped TiO<sub>2</sub> (b) Pure TiO<sub>2</sub> and different concentrations of Nd<sup>3+</sup>-doped TiO<sub>2</sub> (c) Pure TiO<sub>2</sub> and different concentrations of La<sup>3+</sup>-doped TiO<sub>2</sub>, (d) UV-vis diffuse reflectance spectra of pure TiO<sub>2</sub> and different concentrations of Gd<sup>3+</sup>-doped TiO<sub>2</sub>

### 3.4. Optical properties of photocatalysts

Figure 7 (a), (b), (c), (d) shows the UV-Vis DRS spectra of pure TiO<sub>2</sub> and Sm<sup>3+</sup>, Nd<sup>3+</sup>, La<sup>3+</sup> and Gd<sup>3+</sup> ion doping, the absorption spectra can be used to measure the absorption ability of the photocatalysts to different wavelengths of light, calculate the absorption sideband wavelengths of

different photocatalysts, and use the Khan formula (Pal and Krysch, 2016) to find the forbidden band width of the samples, as shown in Table 2.

**Table 2.** Absorption sidebands and forbidden band widths of pure TiO<sub>2</sub> and different concentrations of Sm<sup>3+</sup>, Nd<sup>3+</sup>, Gd<sup>3+</sup> and La<sup>3+</sup> doped photocatalysts

Sample	$\lambda/nm$	$E_g$	Sample	$\lambda/nm$	$E_g$
Pure TiO <sub>2</sub>	394	3.15	3%Gd-TiO <sub>2</sub>	411	3.02
3%Sm-TiO <sub>2</sub>	410	3.02	2%Gd-TiO <sub>2</sub>	428	2.90
2%Sm-TiO <sub>2</sub>	440	2.81	1%Gd-TiO <sub>2</sub>	433	2.86
1%Sm-TiO <sub>2</sub>	437	2.84	0.5%Gd-TiO <sub>2</sub>	431	2.88
0.5%Sm-TiO <sub>2</sub>	428	2.89	3%La-TiO <sub>2</sub>	446	2.78
3%Nd-TiO <sub>2</sub>	456	2.72	2%La-TiO <sub>2</sub>	442	2.80
2%Nd-TiO <sub>2</sub>	468	2.64	1%La-TiO <sub>2</sub>	438	2.84
1%Nd-TiO <sub>2</sub>	441	2.82	0.5%La-TiO <sub>2</sub>	426	2.91
0.5%Nd-TiO <sub>2</sub>	420	2.95			

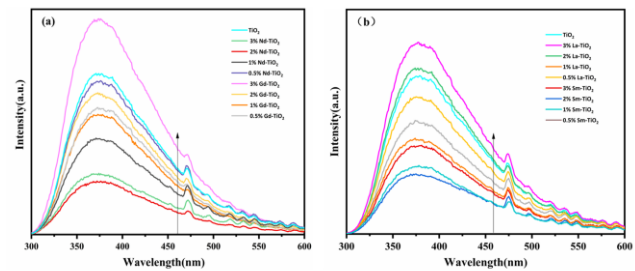
Compared with pure TiO<sub>2</sub>, the doping of Re<sup>3+</sup> (Sm<sup>3+</sup>, Nd<sup>3+</sup>, La<sup>3+</sup>, Gd<sup>3+</sup>) redshifts the absorption band edge of the photocatalyst and enhances the absorption effect in both UV and visible regions, and the forbidden band width of pure TiO<sub>2</sub> is 3.15 eV, which is very close to that reported in the literature (Juan *et al.*, 2022). The forbidden band widths are narrowed after doping with Re<sup>3+</sup>, which can enhance the visible light response of the photocatalyst and thus the photocatalytic activity. The special 4f electron energy level structure of rare earth ions makes the energy values of the ground state and excited state close to each other, and when the 4f electron energy level jumps from the ground state to the excited state, it extends the light absorption wavelength of TiO<sub>2</sub> and enhances the visible light activity of the photocatalyst (Radha *et al.*, 2022). Another reason may be the formation of impurity energy levels between the conduction band and valence band of TiO<sub>2</sub> after the doping of rare earth ions, which changes the relative position of the valence band and conduction band, narrowing the forbidden band width and causing the absorption edge to be red-shifted (Borgarello *et al.*, 1982). The energy difference between the lowest excited state energy level and the ground state energy level in the 4f electron energy levels of different rare earth ions is different, resulting in differences in light absorption ability. 2% Nd-TiO<sub>2</sub> has the most significant light absorption ability in the visible wavelength band greater than 400 nm, and the band gap of the sample is reduced to the greatest extent, with the absorption sideband red-shifted by 74 nm and the corresponding band gap width reduced by 0.51 eV. The above phenomenon may be related to the Nd 4f<sup>4</sup>6s<sup>2</sup> outer electron arrangement, because its outer electron arrangement does not contain d electrons, 4f<sup>4</sup> is easy to interact with O 2p orbitals and form impurity energy levels, which can make the photogenerated electrons jump to form photogenerated carriers in multiple steps at lower energy, improve the visible light response and thus enhance the photocatalytic activity.

### 3.5. Photogenerated carrier separation analysis

Stronger photoluminescence peak intensity means higher photogenerated carrier recombination rate and lower

separation efficiency of photogenerated electrons from holes in photocatalysts (Kumaravel *et al.*, 2020). It can be seen in Figure 8 that the doping of Re<sup>3+</sup> did not change the spectral line shape of TiO<sub>2</sub>, indicating that the introduction of Re<sup>3+</sup> did not cause a new luminescence phenomenon, but only affected the intensity of the PL spectrum, and the samples before and after the modification all showed obvious photogenic spectral peaks in the wavelength range of 350~400 nm.

In Figure 8 (a), the relative intensities of the photogenic spectral peaks are in the order of 3%Gd-TiO<sub>2</sub> > TiO<sub>2</sub> > 0.5%Gd-TiO<sub>2</sub> > 2%Gd-TiO<sub>2</sub> > 0.5%Nd-TiO<sub>2</sub> > 1%Gd-TiO<sub>2</sub> > 1%Nd-TiO<sub>2</sub> > 3%Nd-TiO<sub>2</sub> > 2%Nd-TiO<sub>2</sub>. This means that the separation efficiency of the photogenerated carriers is 3%Gd-TiO<sub>2</sub> < TiO<sub>2</sub> < 0.5%Gd-TiO<sub>2</sub> < 2%Gd-TiO<sub>2</sub> < 0.5%Nd-TiO<sub>2</sub> < 1%Gd-TiO<sub>2</sub> < 1%Nd-TiO<sub>2</sub> < 3%Nd-TiO<sub>2</sub> < 2%Nd-TiO<sub>2</sub>. In Figure 8 (b) the relative intensities of the emission spectra are in the order of 3%La-TiO<sub>2</sub> > 2%La-TiO<sub>2</sub> > TiO<sub>2</sub> > 0.5%La-TiO<sub>2</sub> > 0.5%Sm-TiO<sub>2</sub> > 1%La-TiO<sub>2</sub> > 3%Sm-TiO<sub>2</sub> > 1%Sm-TiO<sub>2</sub> > 2%Sm-TiO<sub>2</sub>, this means that the separation efficiency of the photogenerated carriers is 3%La-TiO<sub>2</sub> < 2%La-TiO<sub>2</sub> < TiO<sub>2</sub> < 0.5%La-TiO<sub>2</sub> < 0.5%Sm-TiO<sub>2</sub> < 1%La-TiO<sub>2</sub> < 3%Sm-TiO<sub>2</sub> < 1%Sm-TiO<sub>2</sub> < 2%Sm-TiO<sub>2</sub>. The above results show that doping with rare earth Re<sup>3+</sup> can help to reduce the complex of photogenerated electrons and enhance the photocatalytic activity of TiO<sub>2</sub>, which is because the introduction of Re<sup>3+</sup> can refine TiO<sub>2</sub> nanoparticles and increase the concentration of oxygen vacancies and defects (Nithyaa and Jaya, 2021). However, when the amount of doping exceeds the optimal ratio, the excess Re<sup>3+</sup> may become the electron complexation center, increasing the electron-hole complexation rate and reducing the photocatalytic activity.

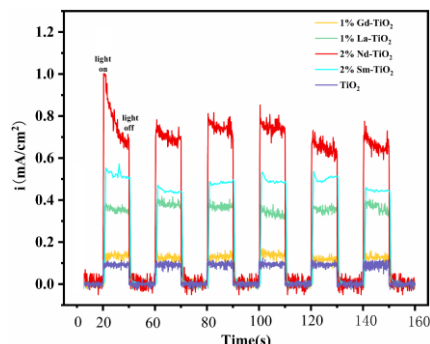


**Figure 8.** (a) Photoluminescence spectra of pure TiO<sub>2</sub>, different concentrations of Nd<sup>3+</sup>-doped TiO<sub>2</sub> and Gd<sup>3+</sup>-doped TiO<sub>2</sub> (b) Photoluminescence spectra of pure TiO<sub>2</sub>, different concentrations of La<sup>3+</sup>-doped TiO<sub>2</sub> and Sm<sup>3+</sup>-doped TiO<sub>2</sub>

Generally speaking, the stronger the photocurrent response, the more efficient the separation of photogenerated electrons from holes, which also means the stronger the photocatalytic activity (Zhu *et al.*, 2020). According to the method of 2.5, pure TiO<sub>2</sub> and 1%Gd-TiO<sub>2</sub>, 1%La-TiO<sub>2</sub>, 2%Nd-TiO<sub>2</sub>, 2%Sm-TiO<sub>2</sub> photocatalysts were selected to prepare working electrodes, and the I-t curves were tested under the conditions of 20 s each of light and dark, as shown in Figure 9, all the five materials had a fast light response, and constant photocurrent values could be obtained without delay in the light phase, and in the the current drops rapidly in the light-avoidance phase and maintains a stable value, indicating that the prepared



electrode materials are sensitive to light response. From the Figure, it can be seen that the photocurrent response of pure  $\text{TiO}_2$  is the weakest, from high to low, 2%Nd- $\text{TiO}_2$ , 2%Sm- $\text{TiO}_2$ , 1%La- $\text{TiO}_2$  and 1%Gd- $\text{TiO}_2$ . This result is consistent with the results of PL spectrum test and degradation test, which indicates that the degradation performance of the photocatalyst is closely related to the carrier separation efficiency.



**Figure 9.** Photocurrent density plots of pure  $\text{TiO}_2$ , Gd- $\text{TiO}_2$ , La- $\text{TiO}_2$ , Nd- $\text{TiO}_2$ , Sm- $\text{TiO}_2$

### 3.6. Photocatalytic activity

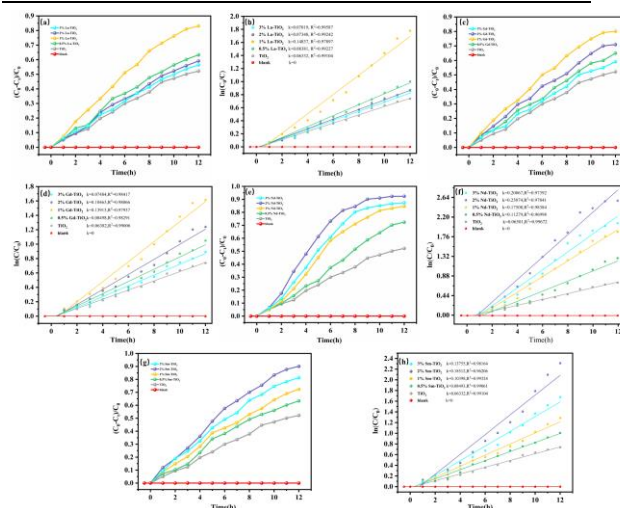
#### 3.6.1. UDMH degradation effect of rare earth modified titanium dioxide

A 350W xenon lamp was used as a simulated light source to degrade UDMH wastewater under simulated sunlight to evaluate the catalytic degradation performance of the samples. The photocatalytic degradation of UDMH wastewater experiment was set up according to 2.4 by adding 3 g of catalyst to the reaction vessel and 50 ml of 100 mg/L UDMH wastewater. As shown in Figure 10 and Table 3, the natural degradation of 100 mg/L UDMH wastewater under simulated sunlight was almost negligible if no photocatalyst was added, and the degradation rate of the prepared pure  $\text{TiO}_2$  to UDMH was 51.1%, and the photocatalytic activities of the samples doped with rare earths were all improved compared with pure  $\text{TiO}_2$ , and the photocatalytic activities were closely related to the amount of rare earth doping. 3%Nd- $\text{TiO}_2$ , 2%Nd- $\text{TiO}_2$ , 1%Nd- $\text{TiO}_2$ , and 0.5%Nd- $\text{TiO}_2$  had 84.5%, 93.3%, 82.1%, and 70.3% degradation rates, respectively, and the doping amount was below 2%, with the increase of  $\text{Nd}^{3+}$  content, the number of traps in the catalyst that could capture photogenerated carriers increased, and the photogenerated electrons and holes could be separated effectively to make the photocatalytic activity enhanced, and the photocatalytic activity is enhanced, and when the  $n(\text{Nd}):n(\text{TiO}_2)$  is greater than 2%, the excess  $\text{Nd}^{3+}$  will form a large number of Ti-O-RE bonds, which will reduce the concentration of surface oxygen vacancies and defects, and will become the complex center of photogenerated carriers (Liang *et al.*, 2020). The photocatalytic activity is reduced, and from the kinetic analysis of the photocatalytic process, the degradation reaction of UDMH follows a pseudo-kinetic first-order model, and the kinetic constants  $k$  values corresponding to the optimal doping amounts of pure  $\text{TiO}_2$  and Nd, Sm, La, and Gd under simulated visible light conditions are related as follows:

2%Nd- $\text{TiO}_2$  ( $k=0.25074\text{min}^{-1}$ ) > 2%Sm- $\text{TiO}_2$  ( $k=0.18512\text{min}^{-1}$ ) > 1%La- $\text{TiO}_2$  ( $k=0.14837\text{min}^{-1}$ ) > 1%Gd- $\text{TiO}_2$  ( $k=0.13913\text{min}^{-1}$ ) >  $\text{TiO}_2$  ( $k=0.06332\text{min}^{-1}$ ). It is obvious that 2%Nd- $\text{TiO}_2$  has the highest kinetic constant, which is consistent with the previous conclusion of photocatalytic activity. Since 2%Nd- $\text{TiO}_2$  exhibited the highest photocatalytic efficiency, all the following experiments were conducted to further investigate the photocatalytic performance. As shown in Figure 11, to further evaluate the mineralization effect of 2%Nd- $\text{TiO}_2$  on UDMH, it was verified by TOC, and the TOC degradation rate of UDMH wastewater could reach 84.8% under simulated sunlight irradiation.

**Table 3.** Degradation rates (DR%), reaction rate constants (K), and linear regression coefficients (R<sup>2</sup>) of pure  $\text{TiO}_2$ , Gd- $\text{TiO}_2$ , La- $\text{TiO}_2$ , Nd- $\text{TiO}_2$ , and Sm- $\text{TiO}_2$  for the degradation of UDMH

Samples	DR%	K(h <sup>-1</sup> )	R <sup>2</sup>
Pure $\text{TiO}_2$	51.1	0.06332	0.99104
3%Sm- $\text{TiO}_2$	80.4	0.13755	0.98164
2%Sm- $\text{TiO}_2$	90.3	0.18512	0.96206
1%Sm- $\text{TiO}_2$	70.9	0.10395	0.99324
0.5%Sm- $\text{TiO}_2$	61.7	0.08493	0.99061
3%Nd- $\text{TiO}_2$	84.5	0.20067	0.97392
2%Nd- $\text{TiO}_2$	93.3	0.25074	0.97841
1%Nd- $\text{TiO}_2$	82.1	0.17508	0.98384
0.5%Nd- $\text{TiO}_2$	70.3	0.11279	0.96998
3%Gd- $\text{TiO}_2$	55.4	0.07484	0.98417
2%Gd- $\text{TiO}_2$	70.3	0.10463	0.98417
1%Gd- $\text{TiO}_2$	79.6	0.13913	0.97937
0.5%Gd- $\text{TiO}_2$	65.4	0.08498	0.98291
3%La- $\text{TiO}_2$	52.3	0.07019	0.99587
2% La- $\text{TiO}_2$	62.3	0.07348	0.99242
1% La- $\text{TiO}_2$	81.1	0.14837	0.97897
0.5% La- $\text{TiO}_2$	60.6	0.08381	0.99227



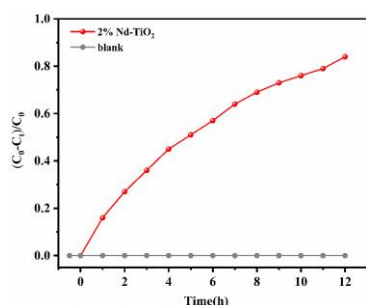
**Figure 10.** Photocatalytic degradation effect and kinetic linear fit of pure  $\text{TiO}_2$ , Gd- $\text{TiO}_2$ , La- $\text{TiO}_2$ , Nd- $\text{TiO}_2$ , Sm- $\text{TiO}_2$

#### 3.6.2. Degradation effect of FDMH and NDMA

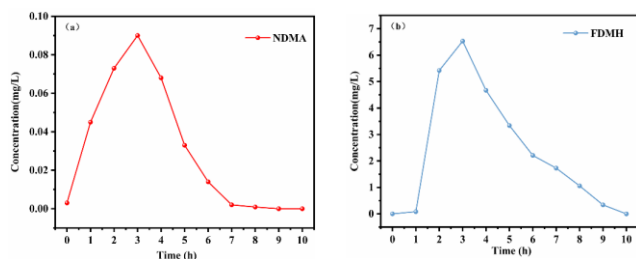
The concentrations of NDMA and FDMH were tested according to the HPLC method previously studied by the group (Zhou *et al.*, 2021). The initial samples contained a small amount of NDMA and FDMH, and the trends of NDMA and FDMH were basically the same, first rapidly increasing and then slowly decreasing, and the rate of



FDMH and NDMA generation by UDMH was greater than the rate of photocatalytic degradation of both at the initial stage, and both NDMA and FDMH reached the maximum concentration around 3h. At this time, the generation rate and degradation rate were almost equal, and the reaction gradually reached After that, the rate of photocatalytic degradation was greater than the rate of generation, and the concentration gradually decreased slowly, and the yield of FDMH was significantly higher than that of NDMA, which might be because FDMH is an intermediate product of UDMH into NDMA, and it is easier to form NDMA from FDMH than from UDMH (Huang *et al.*, 2019), NDMA and FDMH were completely degraded at around 9h and 10h, respectively, indicating that UDMH could be completely mineralized.



**Figure 11.** TOC detection during the degradation of UDMH wastewater

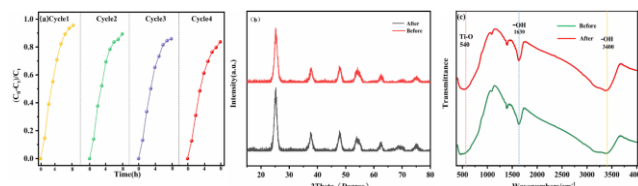


**Figure 12.** Degradation effect of 2%Nd-TiO<sub>2</sub> on NDMA and FDMH

### 3.6.3. Stability of photocatalysts

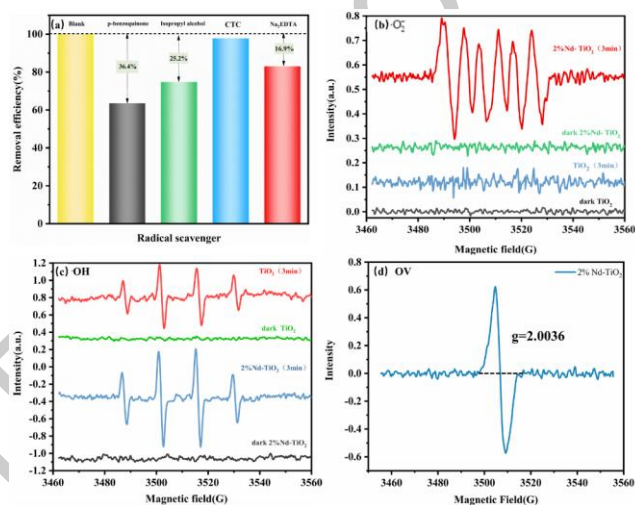
In addition to the photocatalytic activity, the stability of the photocatalyst is also an important evaluation criterion in the process of practical application. 50 ml of UDMH wastewater with a concentration of 50 mg/L was configured, and the recovery cycle experiment was conducted with 2%Nd-TiO<sub>2</sub> under the irradiation of 350 W xenon lamp, and the supernatant was decanted at the end of each cycle, centrifuged and rinsed with deionized water and anhydrous ethanol, dried at 70°C and then recirculated, as shown in Figure 12 (a) shows, the degradation efficiency of four cycles for UDMH was 94.55%, 89.36%, 87.86% and 85.7% in order to further confirm the stability of 2%Nd-TiO<sub>2</sub>, the 2%Nd-TiO<sub>2</sub> after UDMH cycle test was characterized by XRD and FTIR spectra, the results of XRD showed that the crystal structure of 2%Nd-TiO<sub>2</sub> did not significantly changed, and the FTIR results showed that the functional groups of 2%Nd-TiO<sub>2</sub> did not change significantly, and the photocatalytic activity of 2%Nd-TiO<sub>2</sub> was observed to decrease by 8.85% after four cycles, probably due to partial mass loss during centrifugation, washing and

drying, rather than due to its passivation, and 2%Nd-TiO<sub>2</sub> has good stability as an efficient visible photocatalyst, but it also indicates that the powdered catalyst has the defects of tedious operation and easy loss in recoverability.



**Figure 13.** (a) Recycling test results of 2%Nd-TiO<sub>2</sub>, (b) XRD patterns before and after 2%Nd-TiO<sub>2</sub> test, (c) FTIR patterns before and after 2%Nd-TiO<sub>2</sub> test

### 3.7. Active substance identification



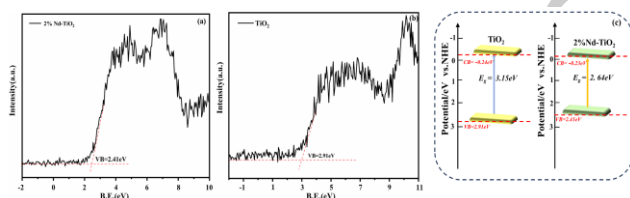
**Figure 14.** (a) active substance capture experiments based on visible light conditions to degrade UDMH, (b) EPR spectra of  $\cdot\text{O}_2^-$  adducts captured under dark and visible light irradiation for 3 min, (c) EPR spectra of  $\cdot\text{OH}$  adducts captured under dark and visible light irradiation for 3 min, (d) EPR spectra of oxygen vacancies of 2%Nd-TiO<sub>2</sub>

The experimental conditions were designed as described in 2.6, 2%Nd-TiO<sub>2</sub> was selected as the photocatalyst, and the degradation rate of UDMH decreased in all groups after adding the trapping agent, indicating that,  $\text{OH}\cdot$ ,  $\text{O}_2^{\cdot-}$ , photogenerated  $\text{h}^+$  and photogenerated  $\text{e}^-$  participated in the photocatalytic degradation process, The values of the effect of isopropyl alcohol, p-benzoquinone, Na<sub>2</sub>EDTA, and carbon tetrachloride on the degradation efficiency of UDMH were 36.4 %, 25.2 %, 16.9 %, and 1.23 %, in that order. Therefore, the photocatalytic effect of active species on UDMH diminishes in the order of  $\text{O}_2^{\cdot-} > \text{OH}\cdot > \text{h}^+ > \text{e}^-$ . The effect of photogenerated  $\text{e}^-$  on UDMH is small and can be ignored,  $\text{O}_2^{\cdot-}$  and  $\text{OH}\cdot$  being the main active species. The EPR technique was used to identify and detect the  $\cdot\text{O}_2^-$  and  $\cdot\text{OH}$  radicals, and 100 mM DMPO was used as the trapping agent, as shown in Figure 13 (a) and (b) under dark conditions, there was no obvious  $\cdot\text{O}_2^-$  and  $\cdot\text{OH}$  radical signal in the EPR, and after 3 min of light, a four-peak EPR signal of 1:2:2:1 for DMPO- $\cdot\text{OH}$  was observed for both TiO<sub>2</sub> and 2%Nd-TiO<sub>2</sub>, and the DMPO- $\cdot\text{O}_2^-$  six-peak signal, indicating that  $\cdot\text{O}_2^-$  and  $\cdot\text{OH}$  radicals play a major role in

this photocatalytic process, which is consistent with the results of the trapping experiments. 2%Nd-TiO<sub>2</sub> exhibits a stronger ·O<sub>2</sub><sup>-</sup> and ·OH signal compared to TiO<sub>2</sub>, indicating that the concentration of both ·O<sub>2</sub><sup>-</sup> and ·OH increases under visible light excitation after doping,  $g=2.0036$  EPR signal of 2%Nd-TiO<sub>2</sub> in 13(d) indicates the presence of oxygen vacancies in the lattice (Santara *et al.*, 2014), both the increase in the concentration of active species and the generation of oxygen vacancies are beneficial to enhance the photocatalytic activity.

### 3.8. Analysis of photocatalytic degradation mechanism

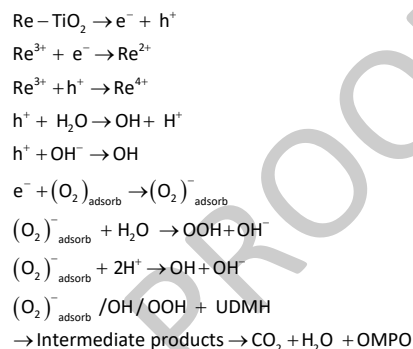
From the analysis of photocatalyst morphological structure, spectral absorption performance and photogenerated carrier separation, it has been confirmed that Re<sup>3+</sup> doping improves the visible light response of TiO<sub>2</sub> and increases the photogenerated electron yield, the energy band edge position of photocatalyst can be calculated by the formula  $E_{CB}=E_{VB}-E_g$ , the energy band structure of TiO<sub>2</sub> and 2% Nd-TiO<sub>2</sub> is shown in Figure 14, combined with the spectral absorption performance and  $E_{VB}=2.91\text{eV}$  for pure TiO<sub>2</sub> measured by XPS spectroscopy,  $E_g=3.15\text{eV}$  from UV-Vis DRS, and  $E_{CB}=-0.24\text{eV}$  calculated.  $E_{VB}=2.41\text{eV}$  for 2% Nd-TiO<sub>2</sub>, UV-Vis DRS gives  $E_g=2.64\text{eV}$ , and  $E_{CB}=-0.23\text{eV}$  is calculated, the  $E_{CB}$  before and after doping are more negative than the oxidation potential of O<sub>2</sub>/·O<sub>2</sub><sup>-</sup> (-0.046 eV vs. NHE), and the photogenerated electrons (e<sup>-</sup>) can react with O<sub>2</sub> to form ·O<sub>2</sub><sup>-</sup>, and the  $E_{VB}$  before and after doping are larger than the reduction potential of H<sub>2</sub>O/·OH (2.4 eV vs. NHE), so h<sup>+</sup> can oxidize H<sub>2</sub>O and OH<sup>-</sup> directly to ·OH (Yang *et al.*, 2022).



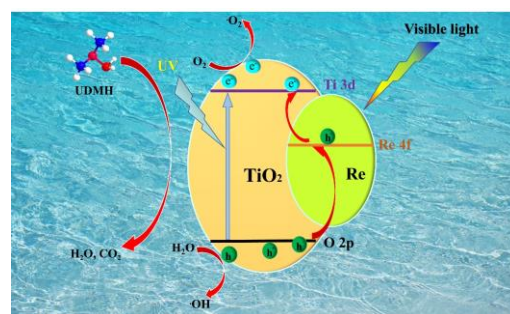
**Figure 15.** (a) VB-XPS spectrum of 2%Nd-TiO<sub>2</sub> (b) VB-XPS spectrum of TiO<sub>2</sub> (c) Schematic diagram of the energy band structure of 2%Nd-TiO<sub>2</sub> and TiO<sub>2</sub>

Figure 15 shows the mechanism diagram of rare earth single doped TiO<sub>2</sub> photocatalytic degradation of UDMH, in pure TiO<sub>2</sub>, under the excitation of UV light, the photocatalyst undergoes electron leap from O 2p to Ti 3d orbitals, Re<sup>3+</sup> doping forms impurity energy level at the bottom of TiO<sub>2</sub> conduction band, which reduces the band gap width and extends the light absorption range to visible light region, under visible light irradiation, photogenerated electrons undergo O 2p to Nd 4f orbital electron leap, the electrons detached from the valence band are more likely to reach the conduction band, producing photogenerated electron-hole pairs, the unfilled 4f orbitals of Re<sup>3+</sup> can capture photogenerated electrons, hindering the recombination of photogenerated electrons and holes, the resulting excited state of Re<sup>3+</sup> can be oxidized with the photogenerated electrons produced in the valence band of TiO<sub>2</sub>, Re<sup>3+</sup> captured photogenerated electrons are reduced to Re<sup>2+</sup>, and The reduction reaction with the hole in the conduction band

generates Re<sup>4+</sup> (Liang *et al.*, 2020), Reaction of photogenerated electrons with O<sub>2</sub> adsorbed on the surface of the receptor, forming (O<sub>2</sub>)<sub>adsorb</sub><sup>-</sup>. ·OH and ·OOH are formed by reacting with H<sub>2</sub>O and H<sup>+</sup>, while the cavity h<sup>+</sup> reacts with OH<sup>-</sup> or H<sub>2</sub>O adsorbed on the surface to form ·OH, and it has been shown that ·OOH, which may be generated during the photocatalytic process, can participate in the UDMH reaction (Wang *et al.*, 1999; Liao and Gurol 1995), The generated (O<sub>2</sub>)<sub>adsorb</sub><sup>-</sup>·OH·OOH etc. are strongly oxidizing and can convert UDMH into CO<sub>2</sub>, H<sub>2</sub>O and other small molecule products. The photocatalytic degradation of UDMH wastewater may involve the following processes:



It has also been reported (Li *et al.*, 2014) that TiO<sub>2</sub> is able to produce Ti<sup>3+</sup> under the effect of photoexcitation and is able to convert the lattice oxygen O<sub>lattice</sub> inside the crystal to surface adsorbed oxygen (O<sub>2</sub>)<sub>adsorb</sub> after the transition state of O (O<sub>2</sub>)<sub>inter</sub>, which in turn can produce oxygen vacancies Ov with (O<sub>2</sub>)<sub>adsorb</sub><sup>2-</sup>, the generated Ti<sup>3+</sup> will produce a positive charge center, and to achieve charge balance, this center will adsorb photogenerated electrons, reducing the recombination efficiency of photogenerated carriers, can adsorb OH<sup>-</sup>, and OH<sup>-</sup> reacts with h<sup>+</sup> to produce more ·OH to participate in the UDMH degradation reaction (El-Bahy *et al.*, 2009), and the generated oxygen vacancies can act as trapping sites for photoexcited electrons and promote the separation of electrons and holes (Ni *et al.*, 2021), It can form an intermediate energy band between the conduction band and valence band of TiO<sub>2</sub> to reduce the band gap and enhance the response of visible light (Zhang *et al.*, 2011), thus improving the photocatalytic activity, and the formation of Ti<sup>3+</sup> and oxygen vacancies on the catalyst surface are favorable to the photocatalytic activity.



**Figure 16.** Schematic diagram of the possible mechanism of degradation of UDMH by Re-TiO<sub>2</sub> under visible light

### 3.9. Analysis of UDMH degradation pathways

The degradation products were studied by GC-MS after 120 min of visible light irradiation, and the details of the identified products have been listed in Table 4. The possible degradation pathways of UDMH are shown in Figure 17. In brief, the degradation pathways of UDMH mainly include bond breaking, condensation, oxidation, and hydrolysis, and UDMH in Figure 16 proceeds along the following transformation pathways from top to bottom: In the first pathway, UDMH is directly converted to (CH<sub>3</sub>)<sub>2</sub>NNH·OOH by the action of free radicals, and then H<sub>2</sub>O is directly removed to generate NDMA, which is further oxidized to dimethylamine and HCHO. In the second pathway, ·OH attacks the N-N bond of UDMH and breaks it to form the dimethylamino radical (CH<sub>3</sub>)<sub>2</sub>N· and the amino radical NH<sub>2</sub>·, and NH<sub>2</sub>· couples with CH<sub>3</sub>· to form methylamine which is then oxidized to formamide, which is further hydrolyzed to form NH<sub>3</sub> and HCOOH, and the dimethylamino radical (CH<sub>3</sub>)<sub>2</sub>N· reacts with the formyl HCO· to form dimethylformamide, and further react to form methylamine, HCHO and NO<sub>2</sub>. In the third path, ·OH attacks the C-N bond of UDMH and breaks it to form CH<sub>3</sub>· and methylhydrazine radical, which is oxidized to form methanol and further oxidized to form CH<sub>2</sub>OH, which is coupled with CH<sub>3</sub>· to form ethanol, and methylhydrazine radical ·CH<sub>2</sub>NNH<sub>2</sub> is oxidized to formylhydrazine radical (H<sub>2</sub>CO·)(CH<sub>3</sub>)NNH<sub>2</sub>, which is coupled with CH<sub>3</sub>· to produce methylformylhydrazide, which reacts with formamide to form the cyclic compound 1-methyl-1H-1,2,4-triazole. In the fourth pathway, OH attacks the N-H bond of UDMH,

which is converted to (CH<sub>3</sub>)<sub>2</sub>NN·H and further oxidized to form (CH<sub>3</sub>)<sub>2</sub>N<sup>+</sup>=N<sup>-</sup>, which is dehydrated to form NDMA after reacting with the reactive group ·OOH, which is further oxidized to dimethylamine and HCHO. In the fifth path, UDMH and HCHO undergo condensation reaction to produce FDMH and H<sub>2</sub>O directly, FDMH is oxidized to produce NDMA, NDMA is further oxidized to dimethylamine, HCHO, and in another case, FDMH is oxidized to produce acetaldehyde dimethylhydrazone, acetaldehyde dimethylhydrazone is further oxidized to produce methylamine, HCHO, NO<sub>2</sub>. As the photocatalytic reaction continues, the intermediate is further mineralized to produce NH<sub>4</sub><sup>+</sup>, NO<sub>3</sub><sup>-</sup>, CO<sub>2</sub> and H<sub>2</sub>O.

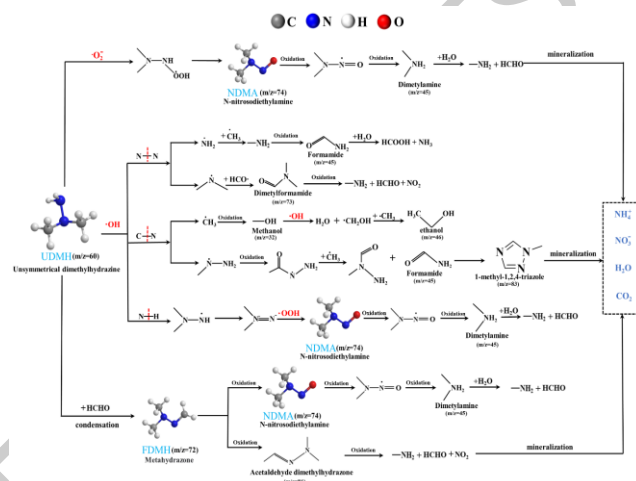


Figure 17. Possible pathways of degradation of UDMH under visible light

Table 4. The intermediate products for the degradation of UDMH

Products*	Molecular formula	Measured mass (m/z)	CAS	Chemical structure
methylenhydrazine	C3H8N2	72	2035-89-4	
Methanol	CH4O	32	67-56-1	
Dimethylamine	CH7N	45	124-40-3	
Ethanol	C2H6O	46	64-17-5	
Acetaldehyde dimethylhydrazone	C4H10N2	86	7422-90-4	
Formamide	CH3NO	45	75-12-7	
1-methyl-1,2,4-triazole	C3H5N3	83	6086-21-1	
N-nitrosodiethylamine	C2H6N2O	74	62-75-9	
Dimethylformamide	C3H7NO	73	68-12-2	

\*Ranked by peak intensity.

## 4. Conclusion

In summary, a series of mesoporous pure TiO<sub>2</sub> and Re<sup>3+</sup>(Sm<sup>3+</sup>, Nd<sup>3+</sup>, La<sup>3+</sup>, Gd<sup>3+</sup>)-doped TiO<sub>2</sub> nanoparticles were

prepared by a simple hydrothermal method, and the photocatalytic activity of these samples was investigated by degradation of UDMH, and the results showed that rare earth ion doping had a great influence on the



photocatalytic activity, and 2%Nd-TiO<sub>2</sub> showed the highest photocatalytic efficiency the good photocatalytic degradation effect can be attributed to its smaller grain size, larger specific surface area, enhanced visible light response after doping, and increased number of oxygen vacancies with minimal carrier complex probability, which made it show good degradation performance on UDMH through synergistic effect, and 2%Nd-TiO<sub>2</sub> did not show significant degradation performance after four cycles, indicating that it possesses excellent recyclable performance. In addition, according to radical capture experiments and EPR, the main active species in photocatalytic reactions are O<sub>2</sub><sup>·-</sup> and OH<sup>·</sup>. The intermediates of UDMH were analyzed and the degradation pathways of UDMH were proposed. The photocatalytic degradation of UDMH by Re<sup>3+</sup> (Sm<sup>3+</sup>, Nd<sup>3+</sup>, La<sup>3+</sup>, Gd<sup>3+</sup>)-doped mesoporous TiO<sub>2</sub> is a well-established method.

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