

Recent advances in performance and effect of Zr doping with ZnO thin film sensor in ammonia vapour sensing

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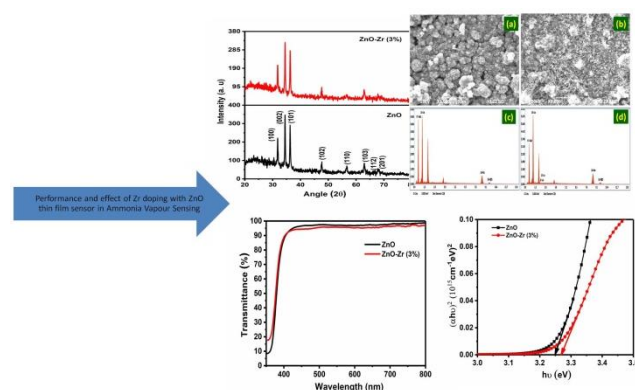
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Graphical abstract



Abstract

Pure and Zr doped ZnO thin films were prepared using SILAR technique. The influence of Zr doping on structural, morphological, optical and gas sensing properties of ZnO has been reported in this study. X-ray diffraction study confirmed the formation of wurtzite structure of ZnO thin film (JCPDS 36-1451) fabricated by SILAR (Successive Ionic Layer by Adsorption and Reaction) technique and the calculated crystallites size of pure and doped ZnO were 39 and 36 nm respectively. SEM analysis of thin films has shown a completely different surface morphology. EDAX spectrum confirmed the presence of different compositional element in the fabricated thin films. Zr (3 wt%) doped ZnO thin film exhibited the best properties with a good transmittance and it has wide band gap of 3.26 eV. Photoluminescence emissions indicated increase in concentration of oxygen vacancies with introduction of dopant. NH₃ vapour sensors were fabricated out of fabricated samples and it was observed that doped samples have significantly high sensing response, good selectivity, fast response and recovery time to ammonia vapour at room temperature.

Keywords: Zr, ZnO, SEM, NH₃, EDAX, SILAR.

1. Introduction

The ammonia gas exposure with high sensitivity have been highly demanded nowadays, since ammonia is a commonly utilized toxic gas in various industrial sectors such as plastics, fertilizers, explosives, textiles, pesticides, food-based, and refrigerant systems and highly toxic and corrosive agent. These agents can threaten human health and environment (Gong *et al.*, 2019; Jayababu *et al.*, 2018; Kwak *et al.*, 2019; Mani and Rayappan, 2015). The metal oxide semiconductors such as SnO₂, ZnO, TiO₂, WO₃, Bi₂O₃, etc based solid state gas sensors are extensively used and it is currently constitute one of the most investigating groups of materials in gas detecting device (Wang *et al.*, 2010). Semiconductor metal oxides based gas sensors have been widely investigated due to their small size, low cost and compatibility with semiconductor fabrication technology. Gas sensors are working on the simple basic principle i.e. the electrical conductivity of a semiconducting metal oxide varies with the composition and the concentration of the gas atmosphere, surrounding it. Amongst various semiconducting metal oxides, ZnO has been extensively studied for the detection of various gases, such as H₂, CO, NH₃, NO₂, and ethanol vapors due to its high mobility of conduction electrons and good chemical and thermal stability under the operating conditions (Hyojin *et al.*, 2010).

ZnO has several applications, such as related to surface acoustic wave devices, piezoelectric devices, varistors, planar optical waveguides, transparent electrodes, UV photo detectors, facial powders, gas sensors, etc. It is an n-type semiconductor of wurtzite structure with direct band gap of about 3.37eV and (Hyojin *et al.*, 2010). It has been demonstrated that the electrical and sensing characteristics of ZnO films can be controlled by doping with various elements or by heat treatment processes

(Dighavkar, 2013; Lukas and Judith, 2007; Karmakar *et al.*, 2007; Norris *et al.*, 2003; Wang *et al.*, 2006; Chao *et al.*, 2019; Singh *et al.*, 2020; Paraguay *et al.*, 2000; Bahsi and Oral, 2007; Al-Hardan *et al.*, 2009; Corlu *et al.*, 2017; Sankar Ganesh *et al.*, 2017; Zhao *et al.*, 2011). The ionic radius of Zr^{+4} (0.8 Å) is larger than Zn^{+2} (0.74 Å). Doping ZnO films with Zr decreases carrier concentration and improves transparency in the visible range (Bharath *et al.*, 2018; Lee *et al.*, 2006). ZnO-based thin films have been prepared by various thin film deposition techniques, such as RF/DC magnetic sputtering deposition, pulsed laser deposition, chemical vapor deposition, chemical bath deposition, SILAR, spray pyrolysis, sol-gel methods, etc. SILAR technique is the most useful method to prepare ZnO thin films. It has the superior advantages such as economical, relatively very simple, high purity reproducibility, does not require high-quality substrates and high-cost equipment than the other methods (Qadri *et al.*, 2000). In this work, polycrystalline semiconductor thin films of pure and Zr-doped ZnO were prepared by SILAR method and the effects of Zr doping on crystallinity, microstructure, surface morphology, optical transparency, photoluminescence were studied and the effect of Zr doping on ZnO and its sensor performance were reported.

2. Experimental

2.1. Preparation of thin film

In this work, pure and 3% Zr-doped ZnO thin films were prepared by Successive Ionic Layer by Adsorption and Reaction method on glass substrates. 3 wt% of Zr sample exhibits the best response and recovery time of 28 and 8 s. Highly pure zinc sulphate ($ZnSO_4$, 99.3 Pure Reagent Grade, Loudwolf) and sodium hydroxide (NaOH, 99.99 % Pure, Merk Pvt Ltd.) were used as a host zinc precursor and oxidizing agents, respectively. Zinc sulphate with 0.1 M concentration was dissolved in deionized water (50 mL) and stirred for 10 min at room temperature to obtain saturated and clear transparent solution with $pH\ 9 \pm 0.2$ and NaOH with 0.2 M concentration and equal amount was added to the prepared solution. This forms the sodium zincate bath and the pH value of the sodium zincate solution is 9 ± 0.2 . Normal SILAR system consists of four beakers, but modified SILAR system has only two beakers (Radhi Devi *et al.*, 2020). The first beaker includes precursor solution and second beaker has hot water. Firstly, a well cleaned glass substrate was vertically immersed into the sodium zincate bath for 20 s. In the second step the substrate was dipped into the double-distilled hot water maintained at 90°C for 10 s to remove the loosely adsorbed ions. The cycles depicted above were performed for a known number of times (80 dipping cycles) to obtain the desired film thickness. In this method we found the pure ZnO thin films. Furthermore, the same procedure was applied to prepare zirconium-doped zinc oxide thin with different doping concentration 3%. The dopant precursor zirconium (IV) oxychloride octahydrate ($Cl_2H_{18}O_9Zr$) was added to a sodium zincate (zinc sulphate + NaOH) solution.

3. Results and discussions

3.1. X-Ray diffraction studies

The crystalline nature of and structure of pure and Zr doped ZnO thin films were identified by X-ray diffraction studies using Bruker (D8 Advance Eco XRD) with Cu K α radiation ($\lambda=1.54056\text{\AA}$). Figure 1 shows the X-Ray diffraction patterns of pure and Zr-doped ZnO thin films. It reveals that ZnO exhibits a hexagonal wurzite structure and is confirmed by the diffraction peaks at (100), (002) and (101) planes and well fit with JCPDS 36-1451. It is observed that the intensities of diffraction peaks decrease with Zr doping. The crystallite size of all the deposited films is calculated by using the very well known Scherrer's formula (Sankar Ganesh *et al.*, 2018).

$$D = \frac{0.9\lambda}{\beta \cos\theta} \quad (1)$$

Where, D is the crystallite size, λ is the wavelength of the X-ray used ($CuK\alpha = 1.54056\text{\AA}$), β is the full width at half maximum intensity and θ is the peak position.

The defects like strain (ϵ) and dislocation density (δ) for the ZnO:Zr was evaluated using the relations

$$\epsilon = \frac{\beta \cot\theta}{4} \quad (2)$$

$$\delta = \frac{1}{D^2} \quad (3)$$

And the number of crystallites (n_c) as

$$n_c = \frac{t}{D^3} \quad (4)$$

where, t is the film thickness.

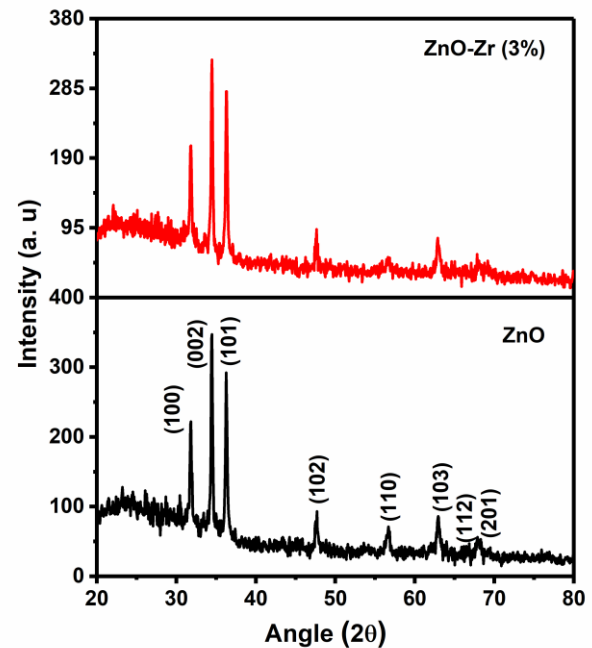


Figure 1. X-Ray diffraction pattern of pure and Zr doped ZnO thin film.

The unit cell volume ' V ' and lattice constants ' a ' and ' c ' of the hexagonal phase of ZnO and Zr doped ZnO films was determined by the following relations (Hong *et al.*, 2019).

$$V = a^2 c (\sin 60^\circ) \quad (5)$$

$$\frac{1}{d^2} = \frac{4}{3} \left\{ \frac{h^2 + hk + k^2}{a^2} \right\} + \left\{ \frac{l^2}{c^2} \right\} \quad (6)$$

Where, (hkl) and d have their usual meanings.

The calculated microstructural parameters of pure and Zr doped ZnO thin films were summarized in Table 1. The average crystallite size was found to be 39 and 36 nm pure and 3 at% Zr doped thin film respectively and noted that Zr-doped ZnO thin films have reduced average crystallite size. It was also noted that the (001) and (002) peaks slightly shifted towards the lower diffraction angle after Zr doping. It has been observed that Zr doping can increase the ' a ' and ' c ' lattice parameters of ZnO films. The ' a ' and ' c ' lattice parameters are 3.005 Å and 5.204 Å for pure ZnO films, and increase gradually to 3.006 Å and 5.206 Å for 3 at% Zr doped ZnO thin films. Zr doping can

increase the lattice parameters of unit cells due to the ionic radius of Zr^{+4} (0.8 Å) is larger than that Zn^{2+} (0.74 Å) (Sankar Ganesh *et al.*, 2018).

3.2. Morphological analysis

The surface morphologies were investigated by (EVO18 CARL ZEISS) Scanning electron microscope (SEM) the image of pure and Zr doped ZnO thin film shown in Figure 2(a-b). The surface morphology of pure ZnO film possesses particles of different sizes with pinhole nature all over the surface of the film and due to doping of 3 wt% Zr, the surface appears with a combination of small and medium-size grains and fibre like structure. The pictures also reveal that Zr doping in ZnO films can markedly reduce the average grain size; this finding agrees with XRD measurements. The SEM images clearly revealed that the influence of the Zr doping in ZnO thin films. EDAX analysis of pure and Zr doped ZnO thin film shown in Figure 3(c-d) confirm the presence of element in the prepared films and Zn, O, and Zr peaks were confirming the existence of Zinc (Zn), Oxygen (O) and (Zr) Zirconium in the prepared thin film. The appearance of un-identified Si and Ca peaks at 1.8 KeV and 3.8 KeV owing to the glass substrate.

Table 1. Microstructural Parameters of pure and Zr doped ZnO thin film

| Micro Structural Parameters | | ZnO | ZnO-Zr (3 wt. %) |
|--|---|---------|------------------|
| Film Thickness (nm) | | 920 | 860 |
| Crystallite size (nm) | | 39 | 36 |
| Dislocation density ($\times 10^{15}$) lines.m ⁻² | | 0.6527 | 0.7706 |
| Strain ($\times 10^{-3}$) | | 2.99 | 3.25 |
| Number of crystallites ($\times 10^{16}$ m ⁻²) | | 1.5343 | 1.8398 |
| Lattice constants(Å) | a | 3.00507 | 3.00621 |
| | c | 5.20493 | 5.20691 |
| Cell volume (Å) ³ | | 40.7055 | 40.7520 |
| Band gap (eV) | | 3.24 | 3.26 |

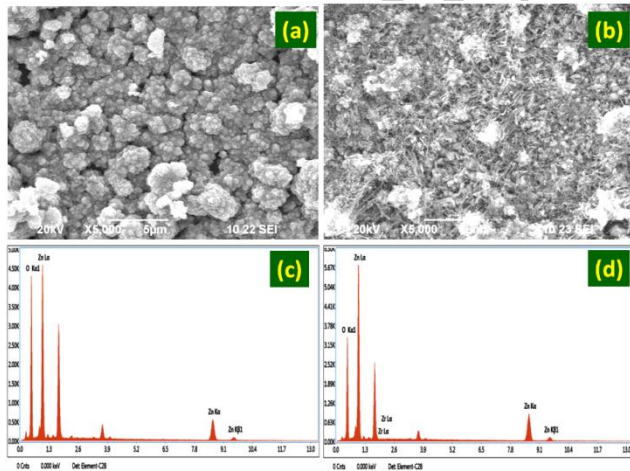


Figure 2. SEM and EDAX Analysis of pure and Zr doped ZnO thin film.

3.3. UV-Vis transmittance analysis

The optical transmittance spectra of the pure and Zr doped ZnO films, with wavelengths from 200 to 800 nm was recorded by using Ocean Optics HR 2000 and shown in Figure 3. It indicated that both pure and Zr-doped films exhibited better transparency. It is observed that Zr

doping change the transparency of the fabricated thin film and it reveals that the surface morphology has a strong influence on the transparency properties of ZnO films (Vijayprasad *et al.*, 2014). Figure 4 shows the Tauc plot of pure and Zr doped ZnO thin films and the band gap values were extrapolated from the straight sections of the plot of $(\alpha h\nu)^2$ versus photo energy $h\nu$ (Vijayprasad *et al.*, 2014). It shows that band gap increase slightly from 3.24 to 3.26 eV for ZnO films doped with Zr and it is due to the effect of average crystallite sizes.

3.4. Photoluminescence analysis

Photoluminescence (PL) spectrum was recorded using Varian Carry Eclipse Photo Luminescence spectrophotometer and observed with 380 nm excitation wavelength. Figure 4 shows photoluminescence spectra of pure and Zr doped ZnO thin film. The emission spectrum consists of sharp peaks at 414, 488, 505, and 529 nm and three small peaks at 440, 447, and 541 nm. The peak intensities decrease with the doping percentage. Different types of intrinsic defects are possible in the ZnO structure which worthy of studying for gas sensing applications (Lee and Park, 2003; Tauc *et al.*, 1966). The peaks 414 nm, 488 and 482 nm are attributed to near band edge (NBE) emission in the UV region due to the recombination of

free excitons and emissions related to zinc interstitial respectively. The peak around 541 nm corresponds to oxygen vacancies (Vo). Further, peak at 447 nm and 505 nm belong to zinc interstitials, oxygen vacancies (Vo) (Nkosi *et al.*, 2020; Patil *et al.*, 2019). Thus Zr doping will create the various defect centers in ZnO thin films that will facilitate the sensing of vapor due to their number of active sites (Figure 5).

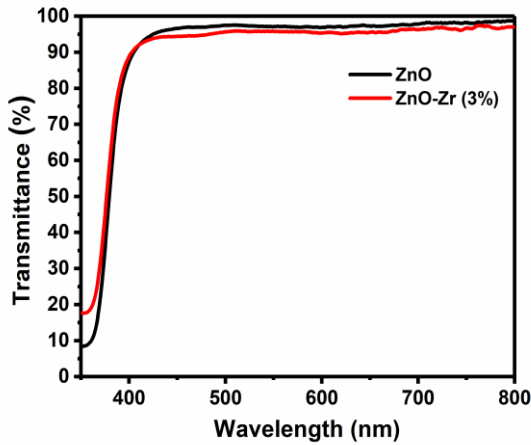


Figure 3. UV Transmittance spectra pure and Zr doped ZnO thin film.

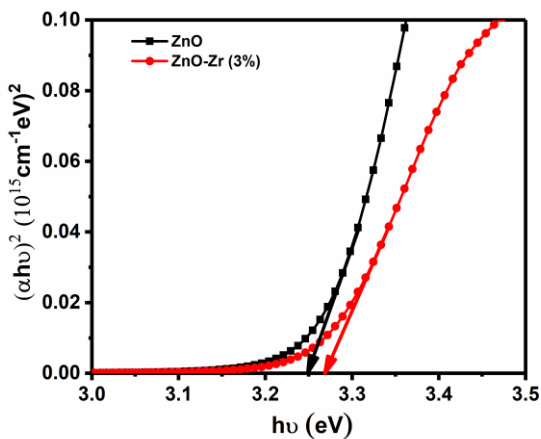


Figure 4. Band gap of pure and Zr doped ZnO thin film.

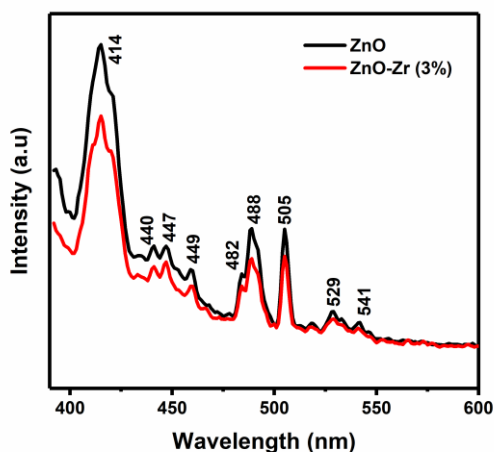


Figure 5. PL spectra of pure and Zr doped ZnO thin film.

Photoluminescence emissions are very important for intrinsic and extrinsic transitions in different type of optical characterization techniques because of its non-destructive nature and ability to yield valuable information (Naushad, 2014; Naushad *et al.*, 2015; Naushad *et al.*, 2015; Naushad *et al.*, 2019; Naushad and ALOthman, 2015).

3.5. NH_3 vapour sensing studies

The home-made setup and a high resistance electrometer (Keithley- 2450) are used to study NH_3 vapor sensing properties of pure and Zr doped ZnO thin films (Anita *et al.*, 2016) and Figure 6 shows the schematic representation of sensor setup used in this study. In the sensing chamber, vapor-producing tube and silver pasted fabricated pure and Zr doped thin film sensor are placed. The electrical contacts made by silver pasted end in the fabricated thin film are connected to the computerized high resistance electrometer (Keithley electrometer 2450, USA). Initially the air resistance was measured in the chamber in the open condition and, the required concentration (ppm) of NH_3 was inserted using the chromatographic syringe. The NH_3 vapor was produced from the tube and is allowed to pass through the fabricated tin film sensor in the chamber. The variation in the resistance was continually recorded with ppm concentration of vapor was introduced into the chamber and also the shift in resistance was calculated after sensing chamber was evacuated. Figure 7 Shows the response and recovery time of deposited pure and Zr doped ZnO thin film on Ammonia vapor sensing at different ppm with time. In this study, pure and Zr doped ZnO thin films were estimated to react to 100 ppm of ammonia by documenting the change in resistance concerning baseline resistance. The vapour sensor system adsorbs or desorbs surface oxygen from pure and Zr doped ZnO thin films and interchanges charge between adsorbed gas, and the thin film layer. This interchange mechanism contributes to changes in the depletion layer and changes in surface or grain boundary. In the experiment, the main reasons for the change in the vapour sensor's electrical properties appear to be the oxygen contribution (Renitta and Vijayalakshmi, 2017; Devi *et al.*, 2020; Krishnan *et al.*, 2017; Shingange *et al.*, 2016; Mani and Rayappan, 2014). In Zr (3 wt. %)-doped ZnO film, the responsivity for NH_3 vapor sensing is more due to the reduction in the contact resistance. Hence, more electrons are injected when ammonia reacts with oxygen species. Therefore, the resistance (R_g) decreased is more compared to the undoped ZnO film. Thus, Zr-doped ZnO films are more sensitive to NH_3 vapor than pure ZnO film. Table 2 shows the NH_3 vapor sensing properties of fabricated thin films. The response time is improved from 59 to 46 seconds and also recovery time reduced from 13 to 6 seconds. Based on the abovementioned reactions, it can be seen clearly that an Zr doping on ZnO can provide more efficient gas adsorption sites and sensitivity improved from 1210 to 1870%. Low-cost and effective materials search for NH_3 vapor sensing is very essential for harmful gases

monitoring. NH₃ is one of the most important chemical for this purpose. However according to Occupational Safety and Health Administration exposure limit of ammonia to humans is 35 ppm for 10 min, beyond this limit NH₃ is harmful for our environment (Pandey, 2016; Pandey *et al.*, 2015; Pandey *et al.*, 2016).

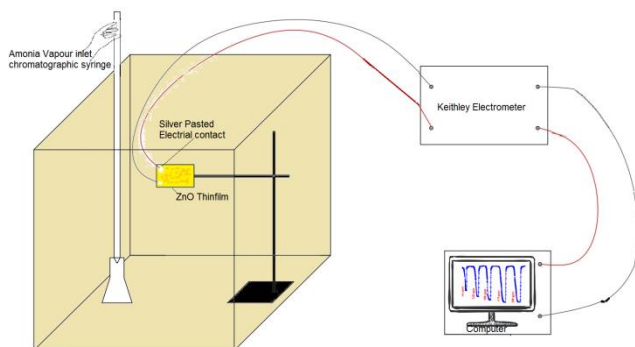


Figure 6. Schematic of Ammonia vapour sensing setup.

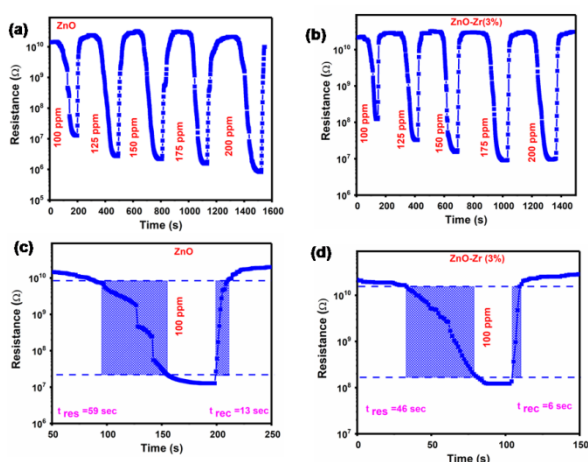


Figure 7. Response and recovery time of deposited pure and Zr doped ZnO thin film on Ammonia vapour sensing.

Table 2. Ammonia vapour sensing properties

| NH ₃ Sensing Properties | ZnO | ZnO-Zr (3%) |
|------------------------------------|------|-------------|
| Response Time (Sec) | 59 | 46 |
| Recovery Time (Sec) | 13 | 6 |
| Sensitivity (%) | 1210 | 1870 |

4. Conclusions

The pure and Zr doped ZnO thin films have been successfully synthesized using the SILAR technique. Both pure and Zr doped ZnO thin films were crystallized with wurtzite structure. It is revealed that the crystallite size decreases from 39 to 36 nm with the doping of Zr 3 wt%. The effect of Zr doping on surface morphology was observed and the elemental composition was confirmed by EDAX analysis. The optical transmission of ZnO films was influenced by Zr addition and the band gap measured from the absorption spectrum is increased from 3.24 to 3.26 eV. Pure and Zr doped ZnO films are showing more than 90% transparency in the visible region. Photoluminescence emissions indicated increase in concentration of oxygen vacancies with introduction of

dopant. An inhouse setup was made to analyse the sensing performance of Pure and Zr doped ZnO thin films. The Zr doping gives more electrons in the ZnO matrix, which helps to improve the quantity of adsorbed oxygen, leading to increased sensitivity and improve the NH₃ vapor sensor efficiency. The fabricated Zr doped ZnO sensor showed good selectivity, fast response and recovery time to NH₃ vapour at room temperature.

Hence our present study indicated Zr (3 wt%) doped ZnO thin film exhibited the best properties with a good transmittance and it has wide band gap of 3.26 eV, whereas, photoluminescence emissions indicated increase in concentration of O₂ vacancies with introduction of dopant.

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